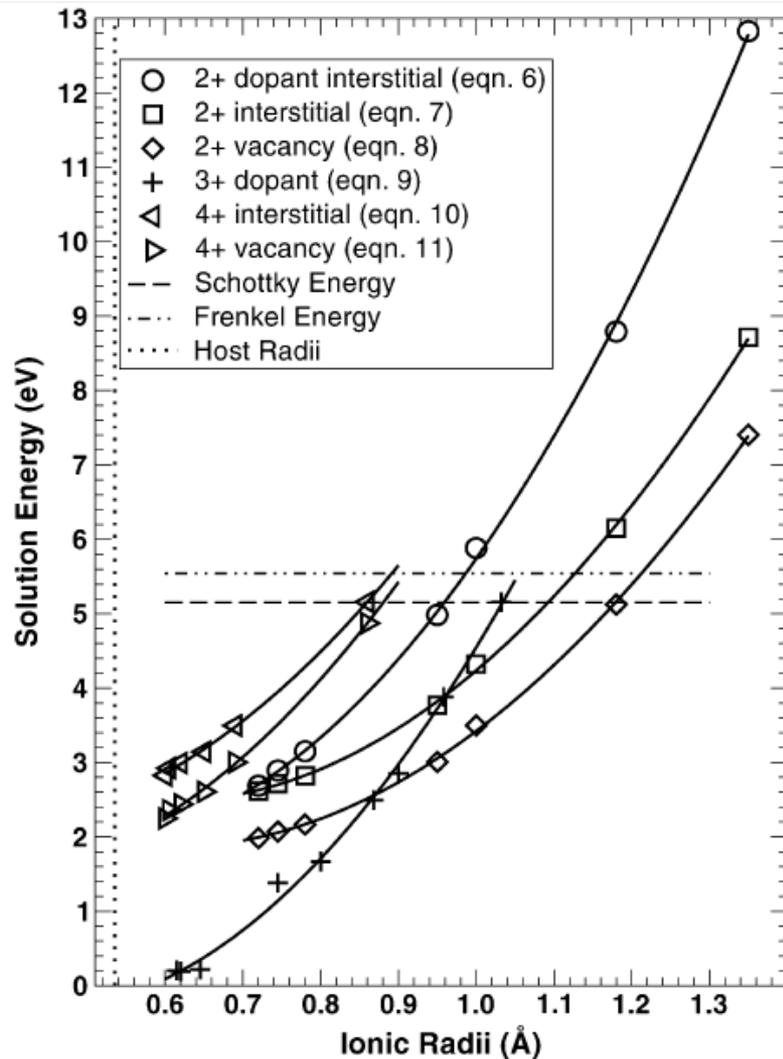


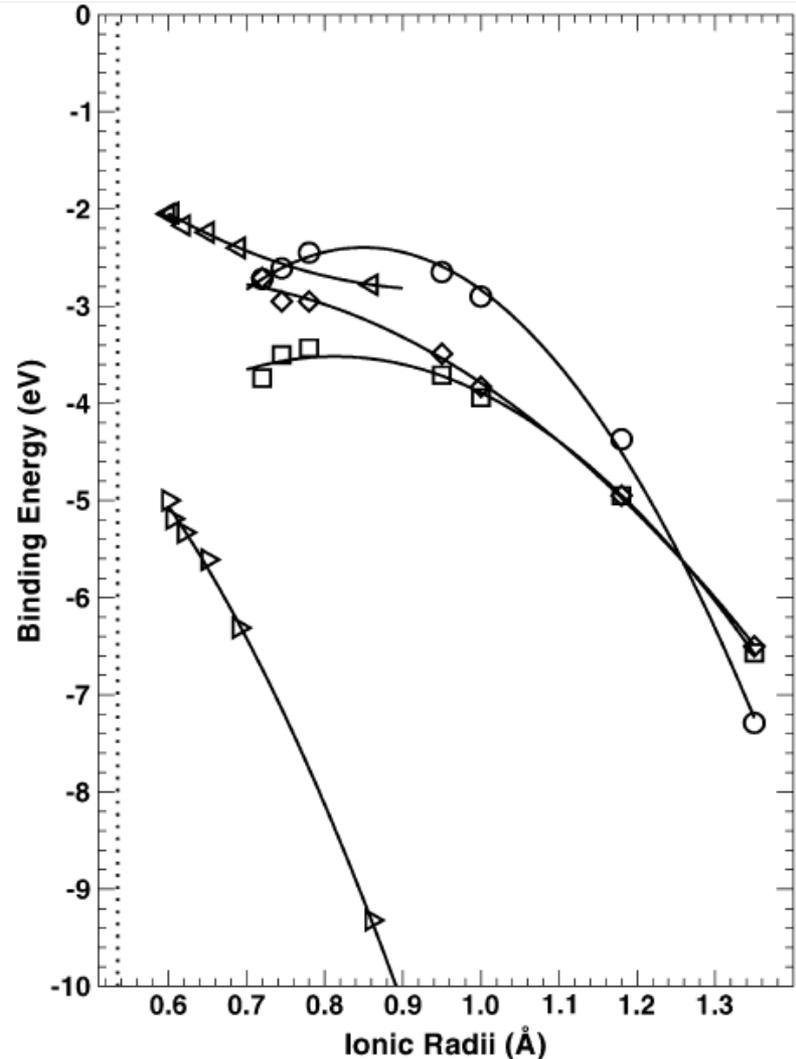
Research in Next 6-12 Months

- G.B. thermal conductance in doped samples
 - Bi crystals
 - Polycrystals
- Continue Diffusion in Al_2O_3 (Cr^{3+} , O^{2-} , 18)
 - Approximate ΔS
- Diffusion in Ni-W
 - Borisov Model $\Delta H_{\text{GB}} \sim \Delta H_{\text{L}} - \gamma$ (w/o complexions)
 - Complement Lehigh/Clemson-UCSD work

Defects in Alumina

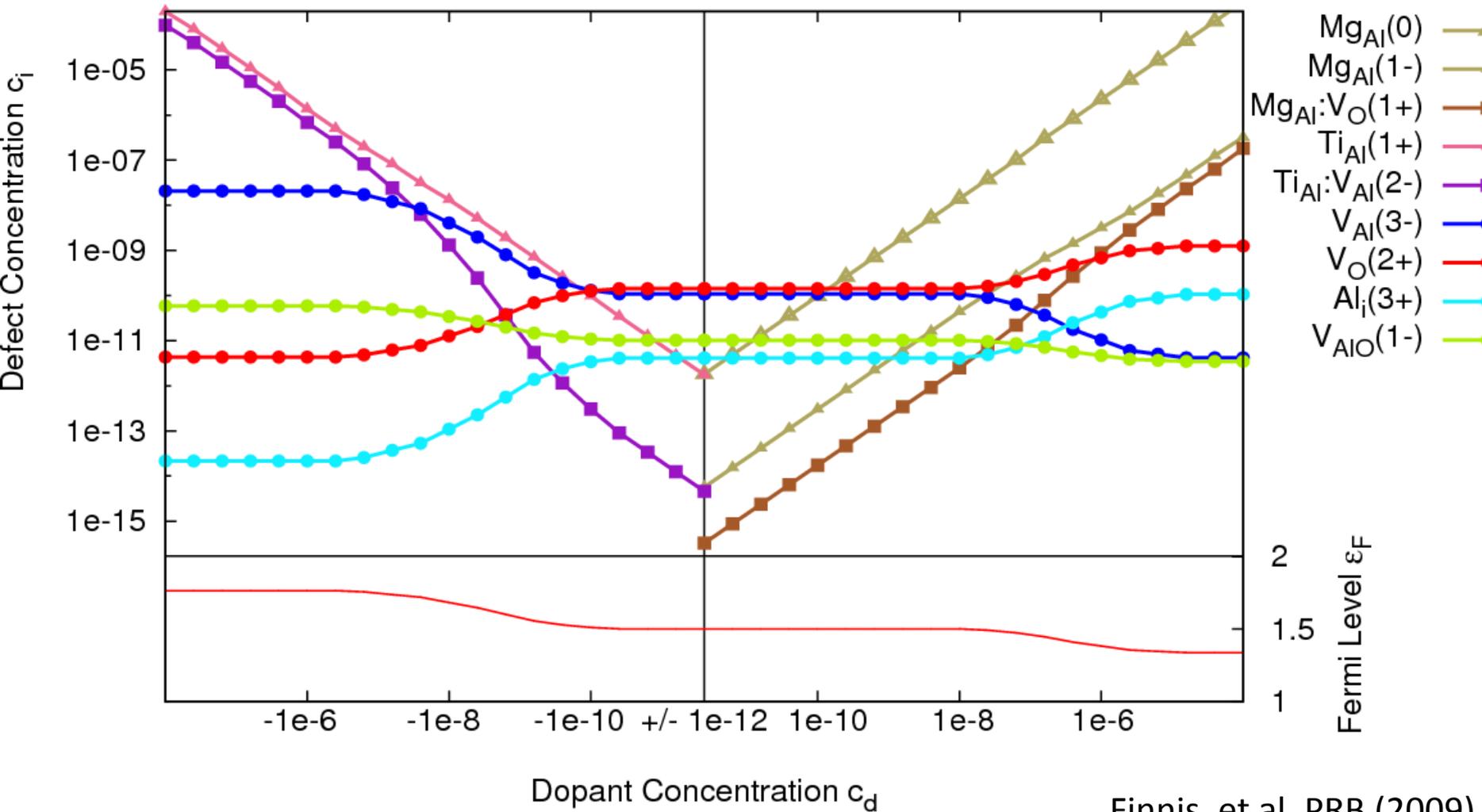


(a) Isolated Solution Energies



(b) Cluster Binding Energies

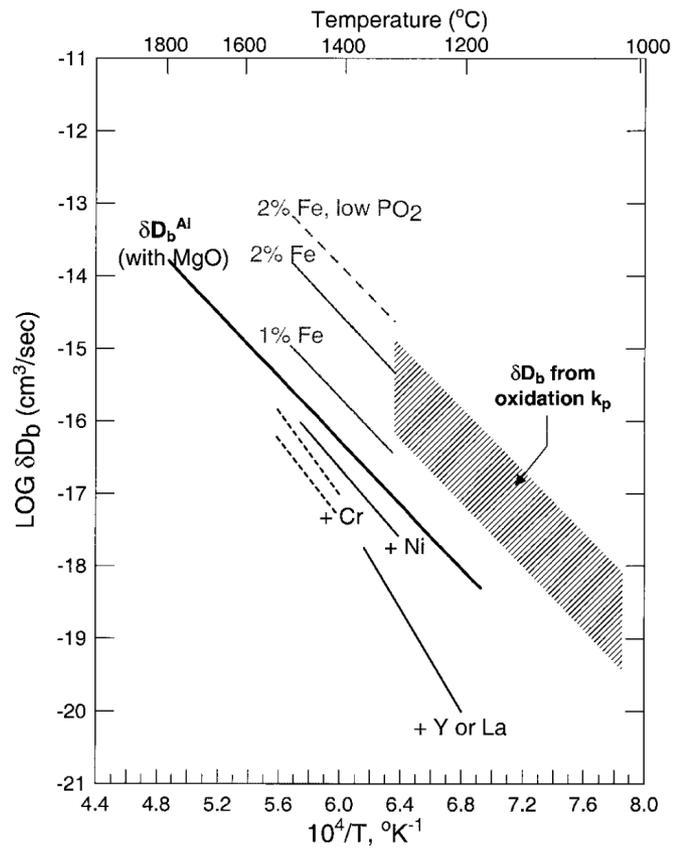
Defects in Alumina



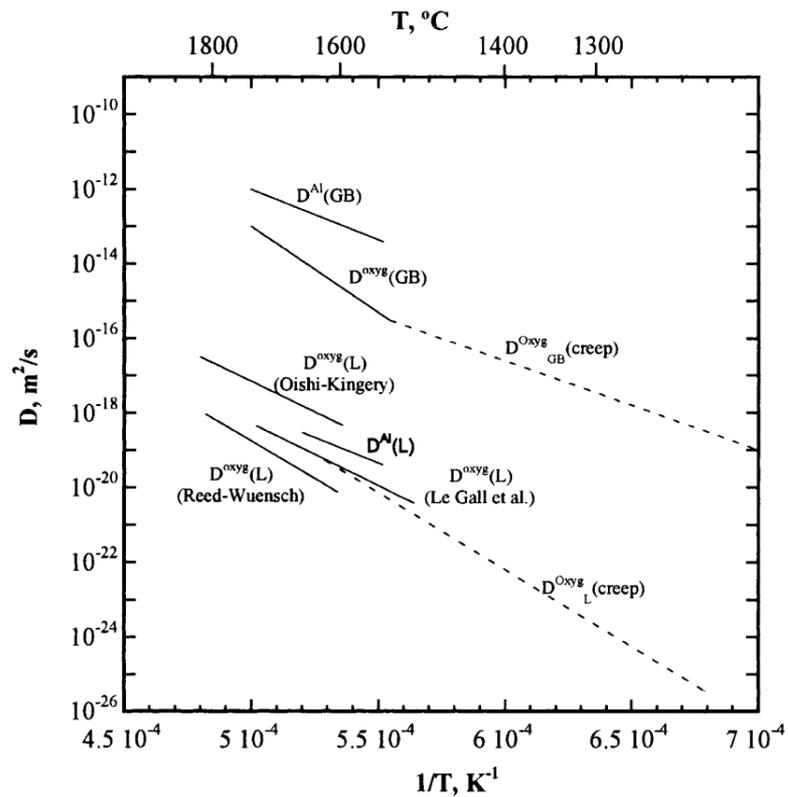
Finnis, et al. PRB (2009)

Fraction of mobile defects in Si^{4+} at 1400 $\sim 10^{-11}$

P. Hou



O. Ruano, J. Wadsworth, and O. Sherby



Alumina Diffusion - Motivation

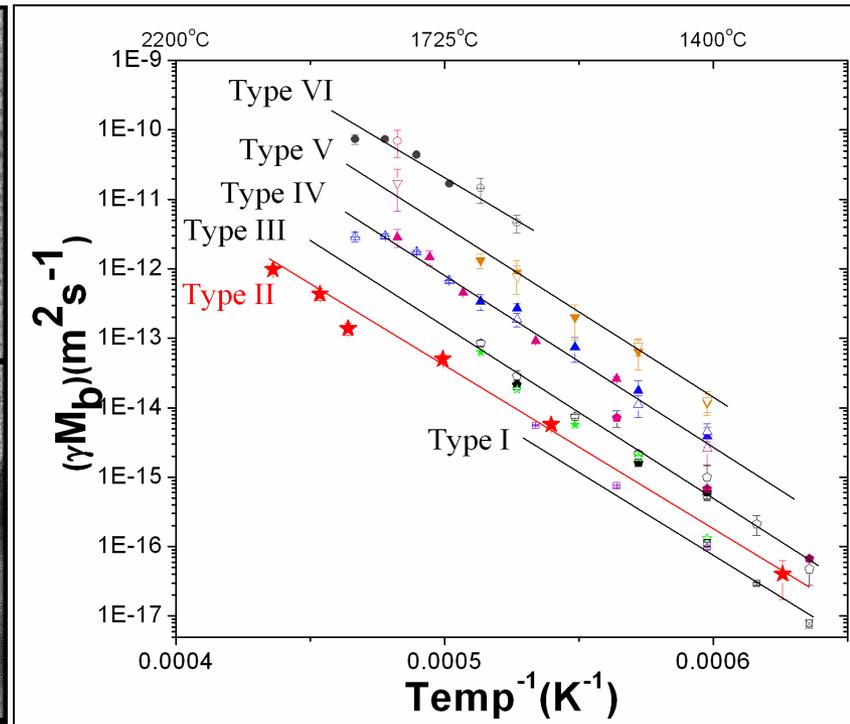
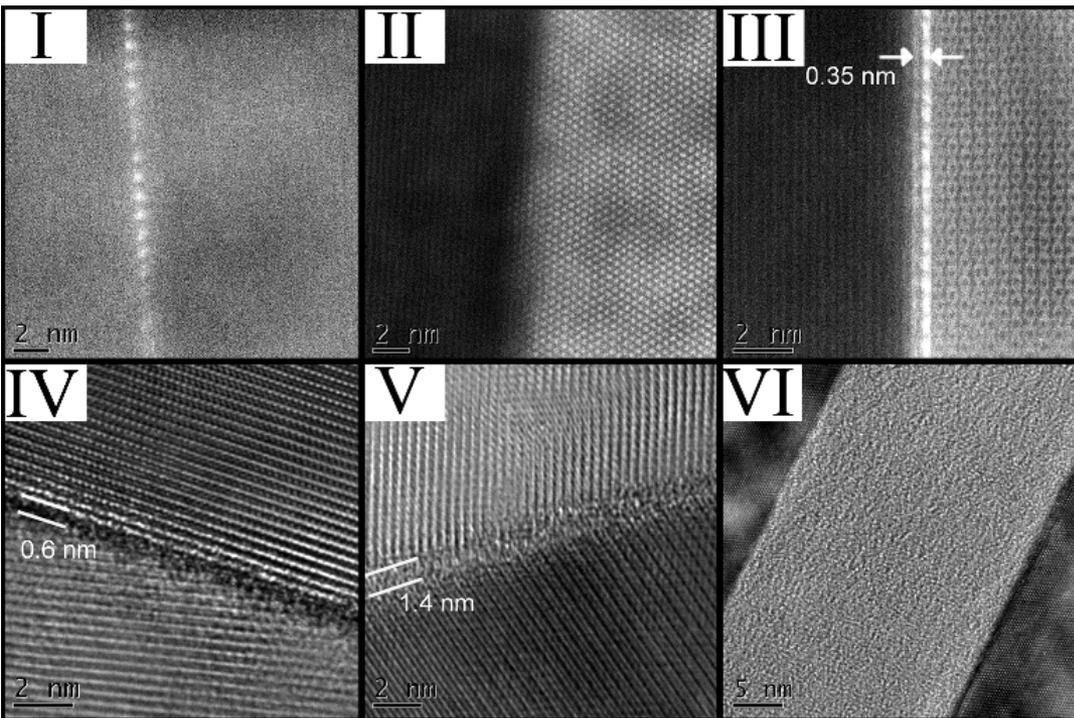
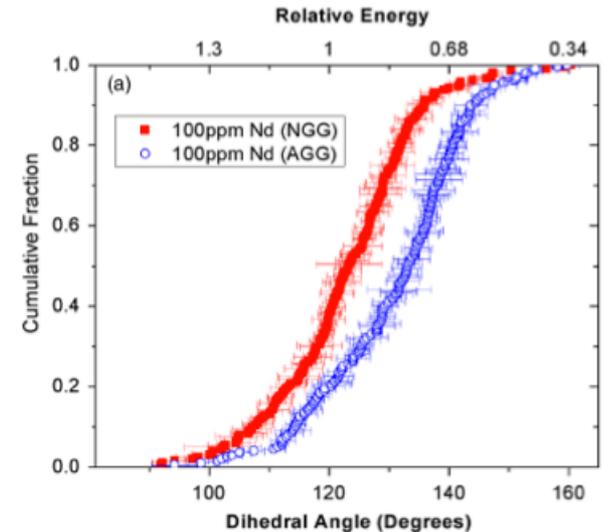
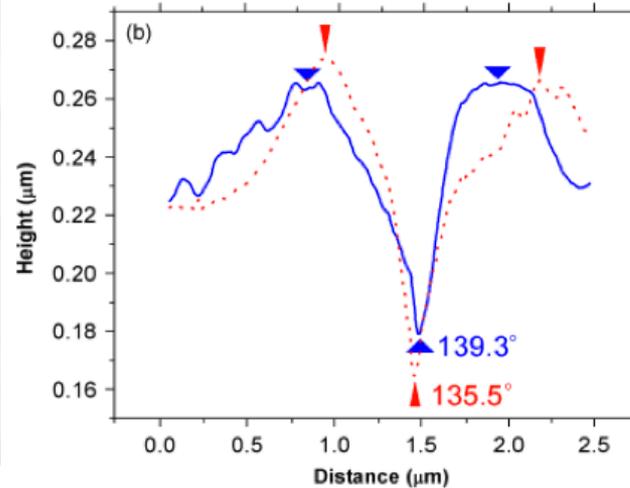
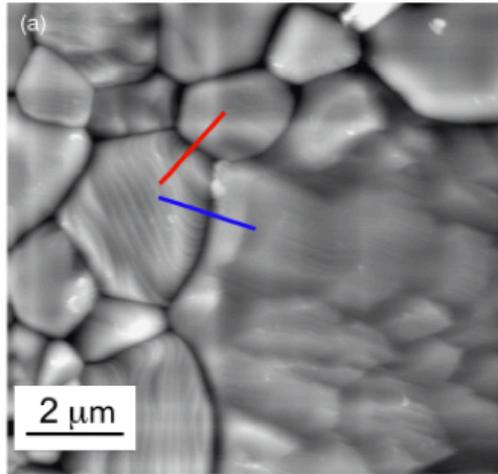


Table I. The Mean Relative Energies of Different Grain-Boundary Complexes Occurring as Normal and Abnormal Grains in Doped and Undoped Alumina Annealed at Different Temperatures

| Chemistry | Temperature (°C) | Complexion | Relative energy | % energy change (complexion transition) |
|--|------------------|------------|-----------------|---|
| Undoped | 1400 | II (NGG) | 1.11 | |
| | 2020 | II (NGG) | 1.08 | |
| 100 ppm-Nd ₂ O ₃ | 1400 | I (NGG) | 0.95 | -16 |
| | 1400 | III (AGG) | 0.8 | |
| 100 ppm-Y ₂ O ₃ | 1400 | I (NGG) | 0.57 | -46 |
| | 1400 | III (AGG) | 0.31 | |
| 500 ppm-MgO | 1400 | I (NGG) | 1.07 | -26 |
| | 1700 | III (NGG) | 0.79 | |
| 30 ppm-CaO | 1200 | I (NGG) | 0.82 | -20 |
| | 1200 | III (AGG) | 0.69 | |
| 200 ppm-SiO ₂ | 1200 | I (NGG) | 0.68 | -10 |
| | 1200 | III (AGG) | 0.61 | |

Relative Grain Boundary Energies

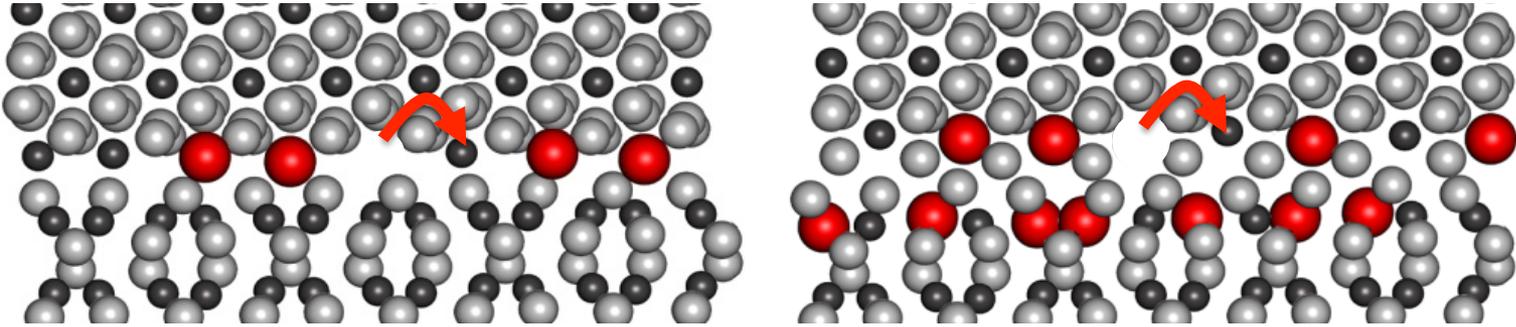


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| 200 ppm-SiO ₂ | 1200 | I (NGG) | 0.68 | -10 |
| | 1200 | III (AGG) | 0.61 | |
| 30 ppm-CaO | 1400 | III (NGG) | 1.02 | 0.1 |
| | 1400 | IV+ (AGG) | 1.02 | |
| 200 ppm-SiO ₂ | 1400 | III (Basal plane) | 0.77 | |
| | 1400 | III (NGG) | 0.65 | 9.5 |
| 200 ppm-SiO ₂ | 1400 | IV (AGG) | 0.71 | |
| | 1750 | IV (NGG) | 0.98 | -1.7 |
| | 1750 | V+ (AGG) | 0.96 | |

Dillon et al. *JACerS* (2012)

Rohrer et al. continuing more detailed analysis

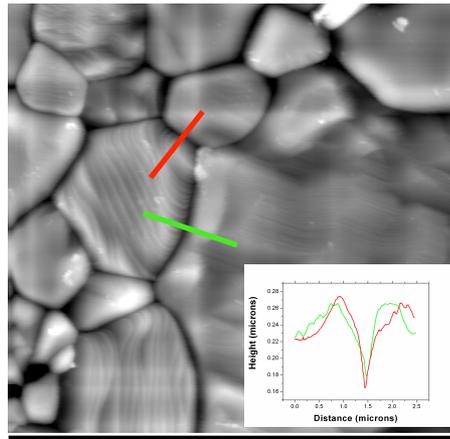
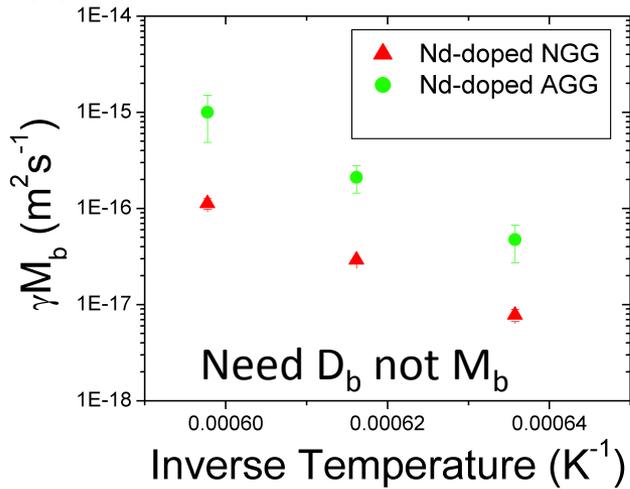
Comparing Complexions (ΔH & ΔS)



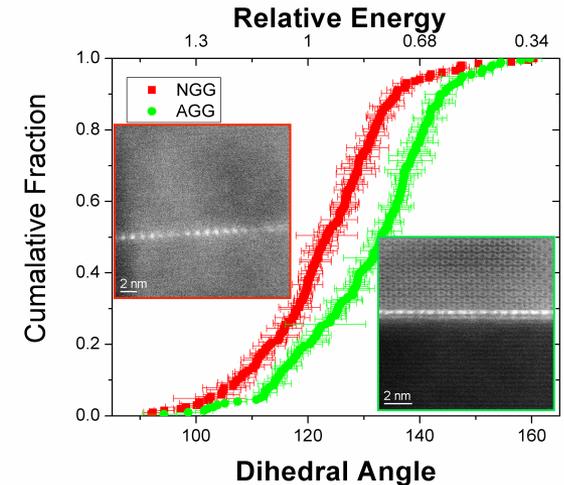
$$\Delta\gamma = \Delta\rho [RT \ln(g a_{\perp}^2 v_{\perp}^* / g \kappa_{\parallel} a_{\parallel}^2 v_{\parallel}^*) - T(\Delta S_{f,\perp} - \Delta S_{f,\parallel}) + (\Delta H_{f,\perp} - \Delta H_{f,\parallel}) + (\Delta H_{m,\perp} - \Delta H_{m,\parallel})]$$

Simplifying Assumptions: $\Delta\gamma = \Delta\rho [-T(\Delta S_{\perp} - \Delta S_{\parallel}) + (\Delta H_{\perp} - \Delta H_{\parallel})]$

Approach:

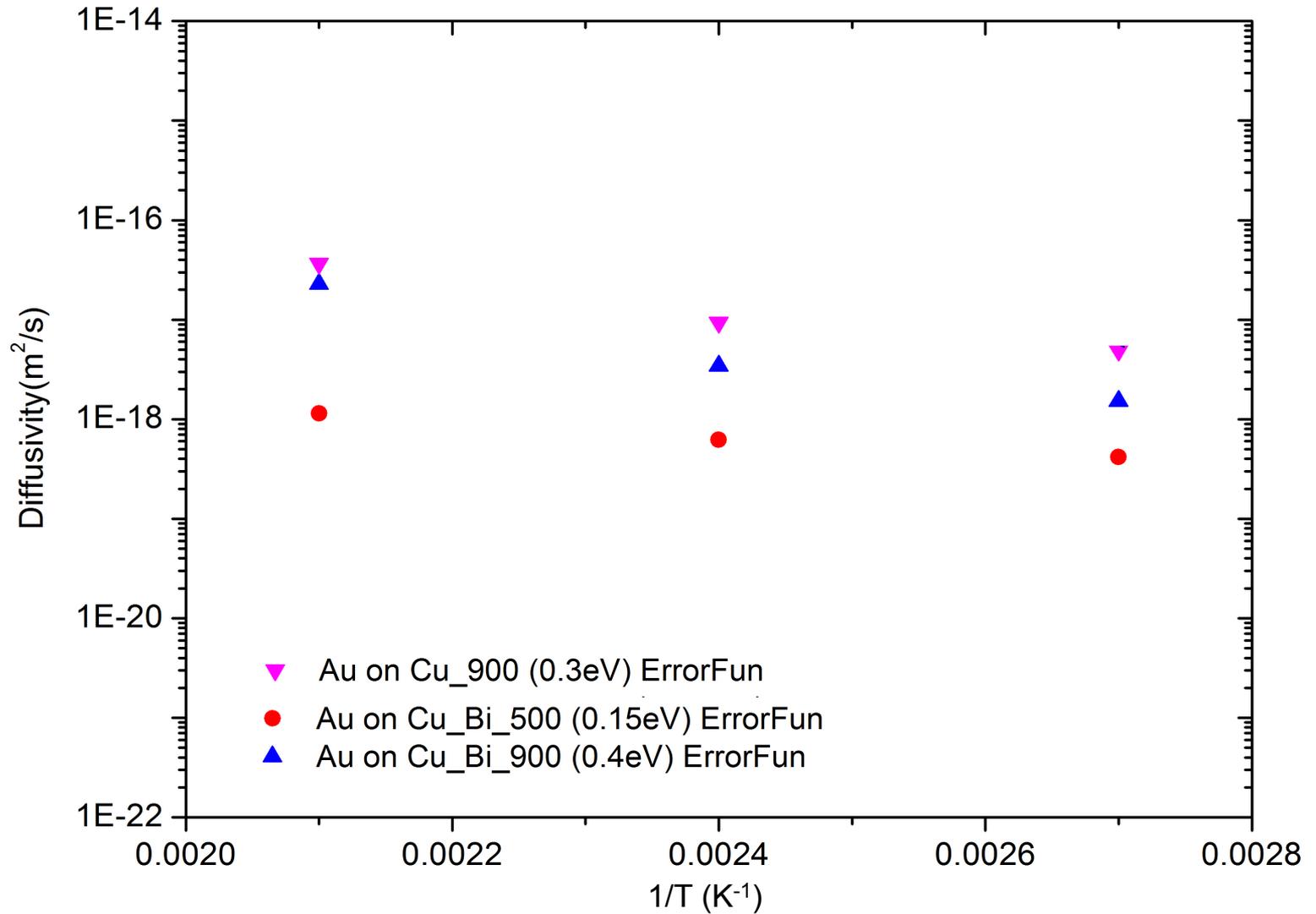


10 micron



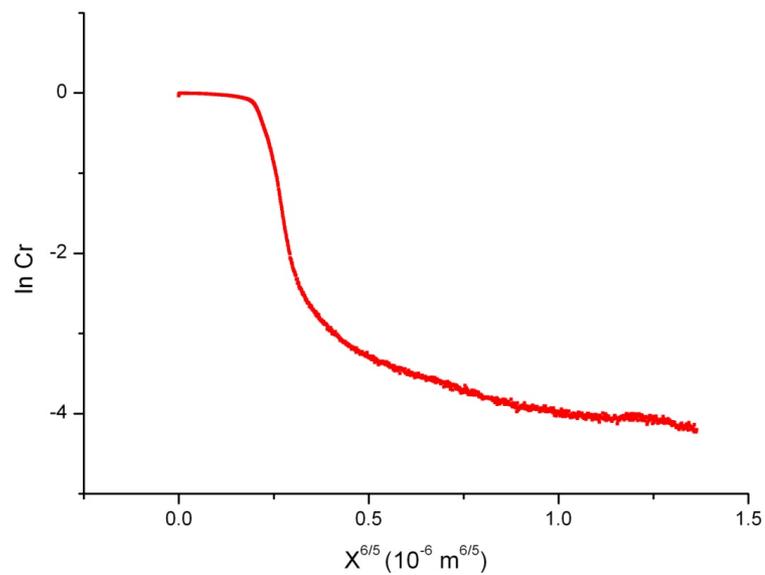
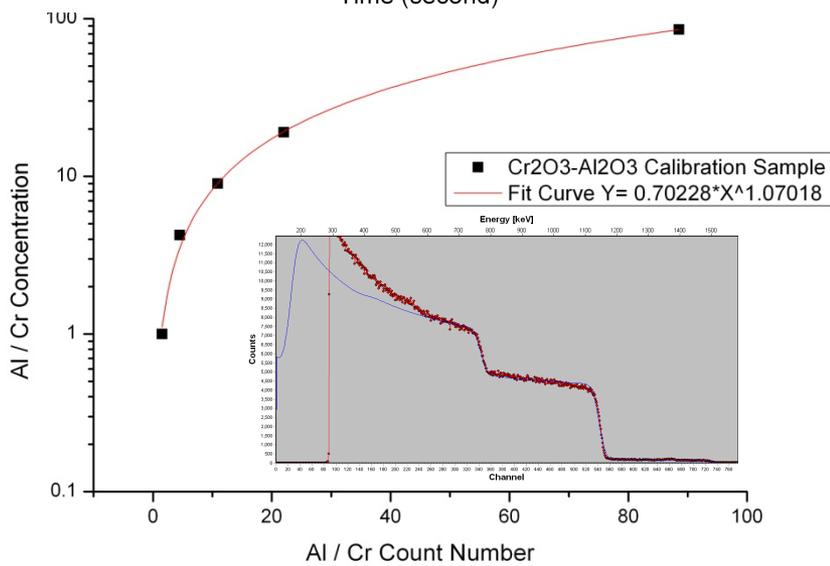
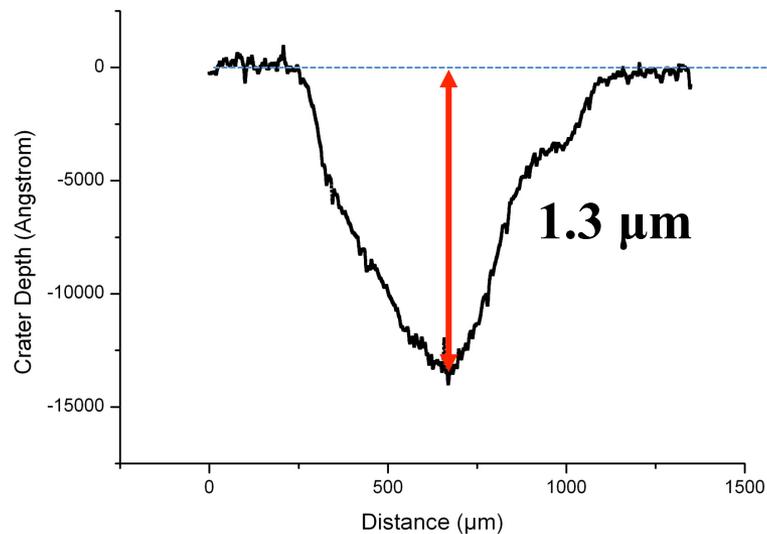
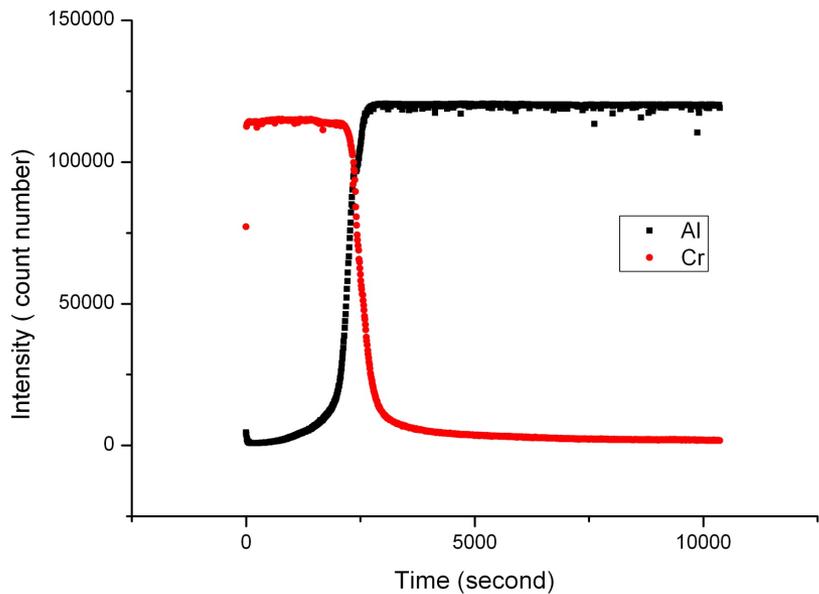
Check: $\gamma = \rho [RT \ln(g a_b^2 v_b^* / g \kappa_a a_l^2 v_l^*) - T(\Delta S_{\perp} - \Delta S_b) + (\Delta H_{\perp} - \Delta H_b) + (P\Delta V_{\perp} - P\Delta V_b)]$

Diffusion in Bi doped Cu

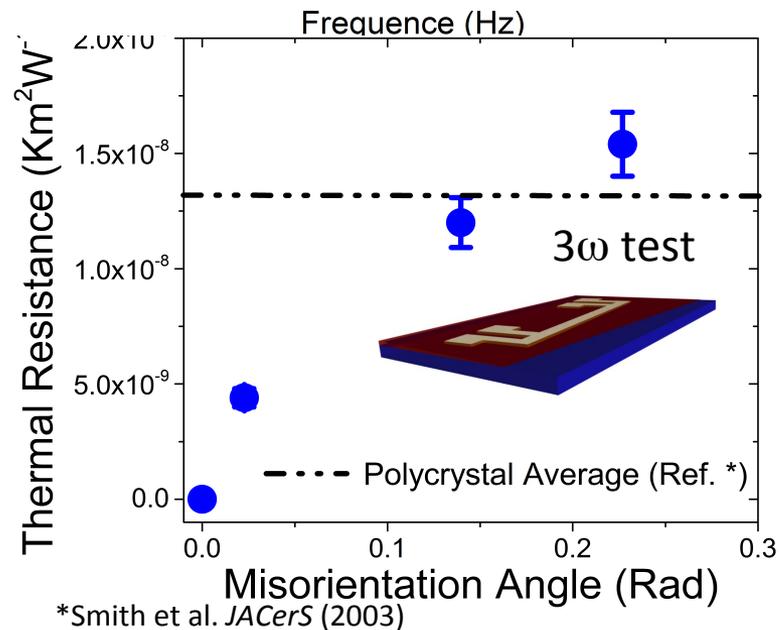
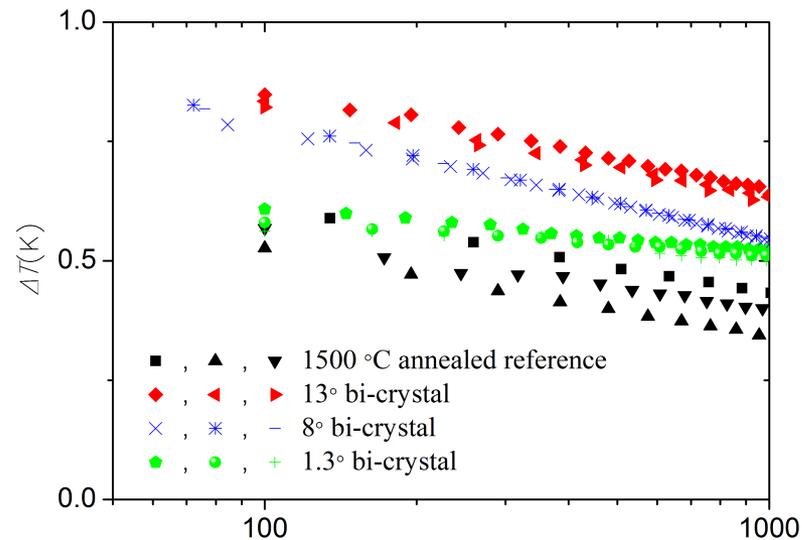
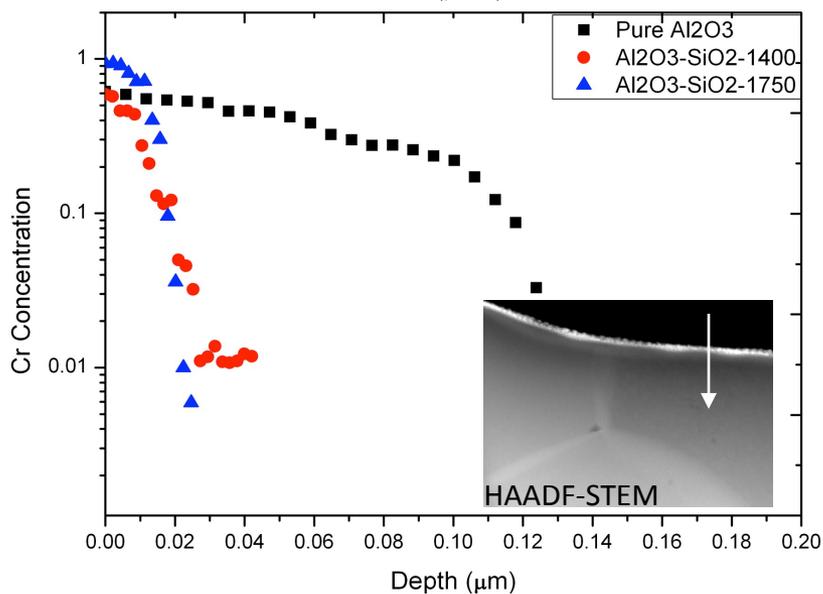
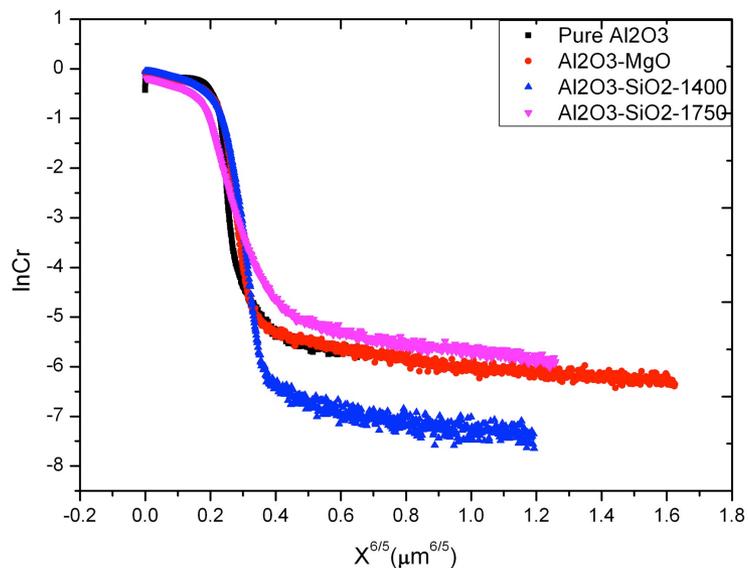


Au diffusivity-temperature dependence in the Cu(Bi) alloys

Cr³⁺ Diffusion in Al₂O₃

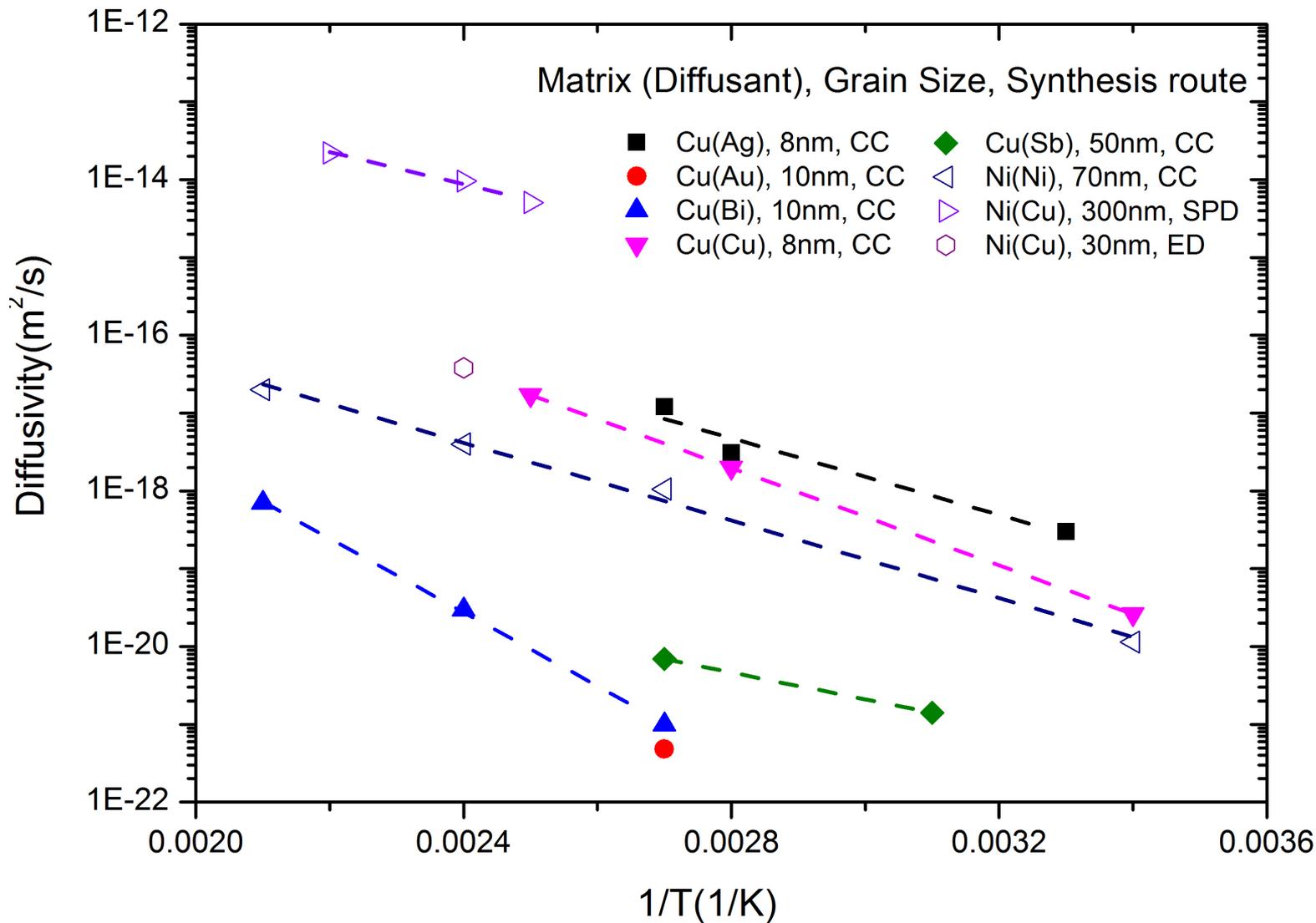


Continued Work: Obtaining D , γ , T_c in Al_2O_3 (then V^*)

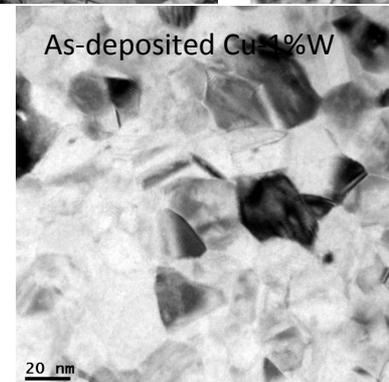
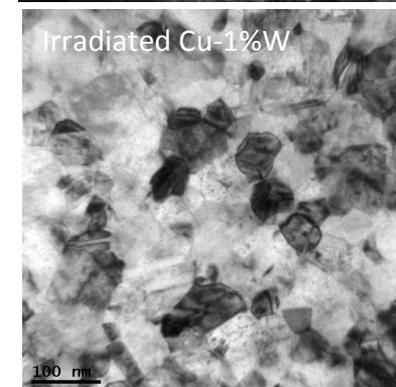
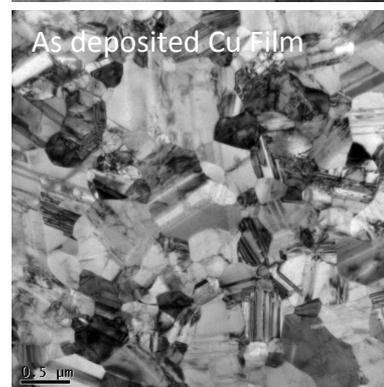
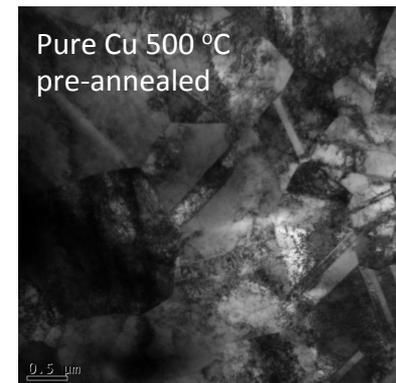
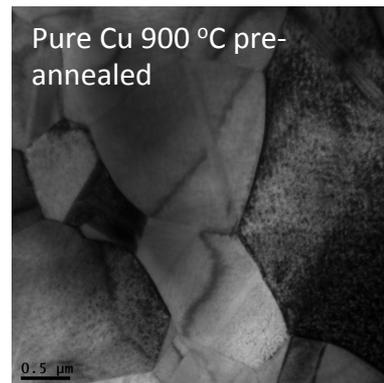
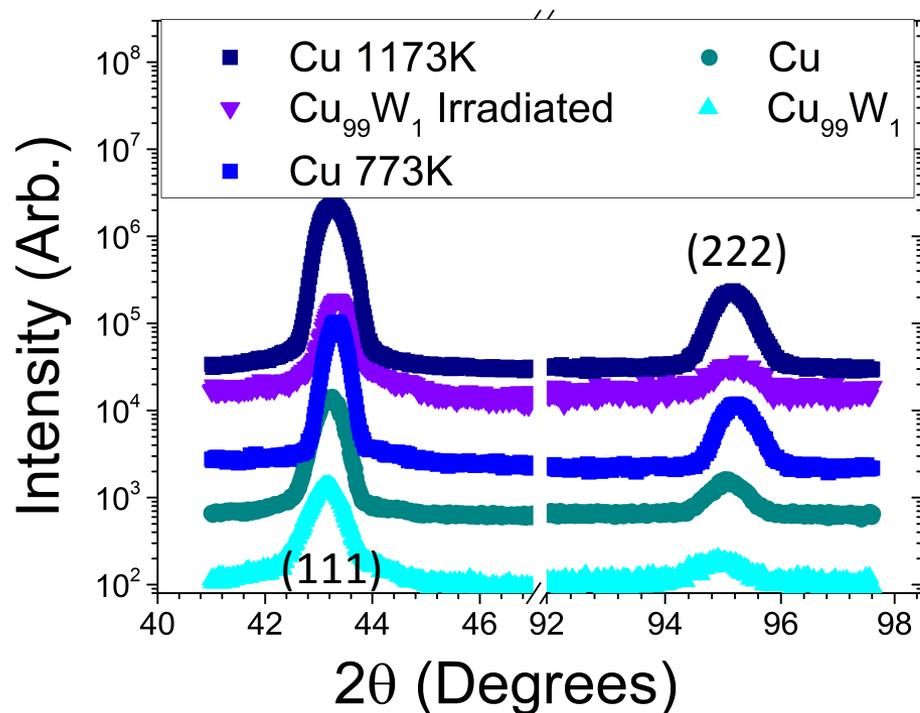


*Smith et al. *JACerS* (2003)

GB Diffusion Nanograin Alloys- Motivation

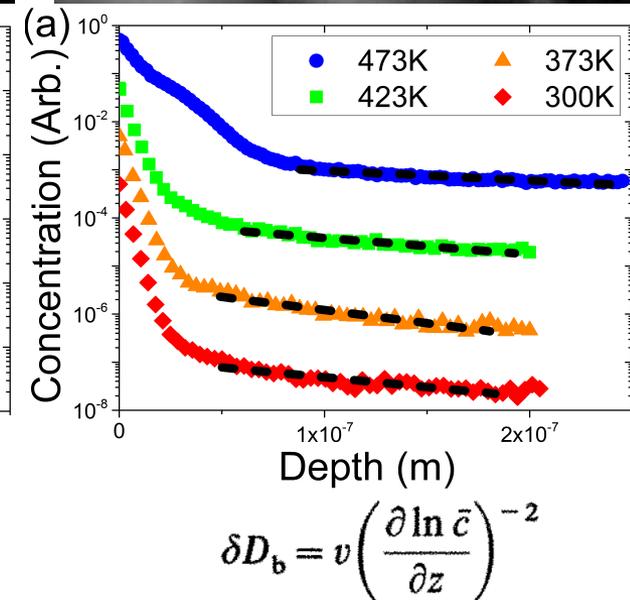
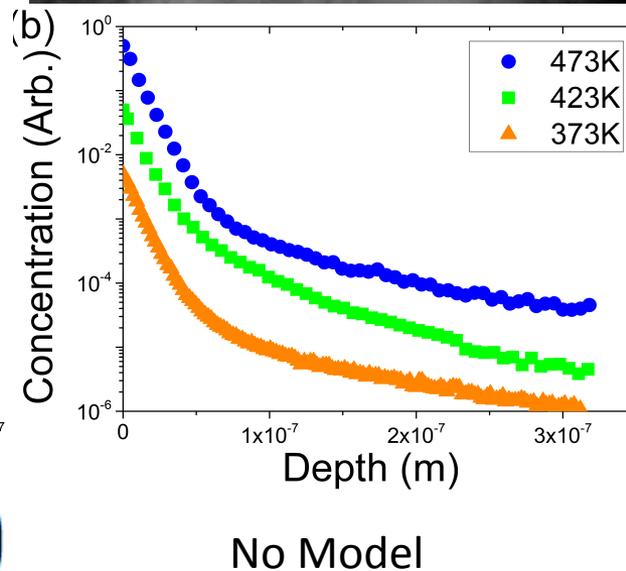
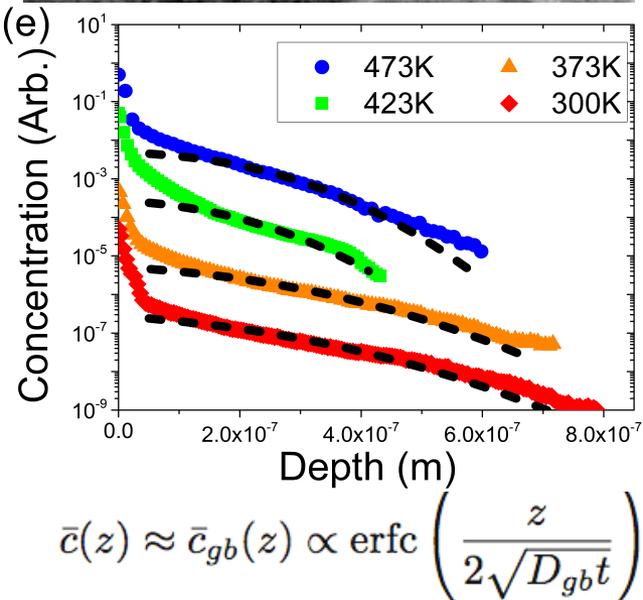
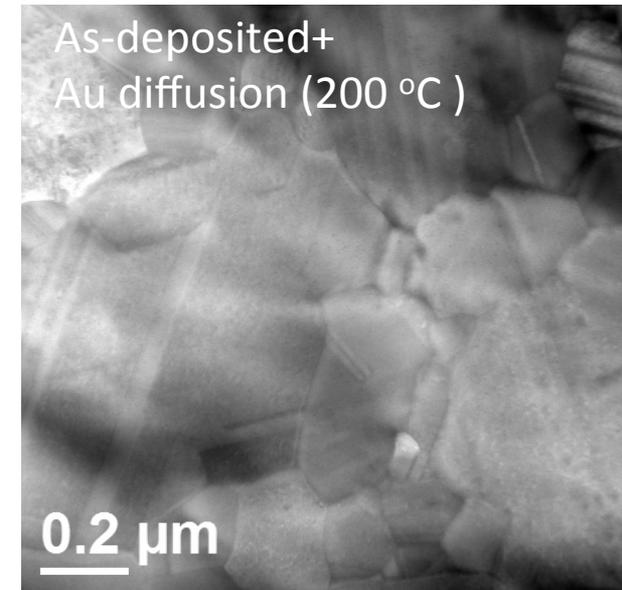
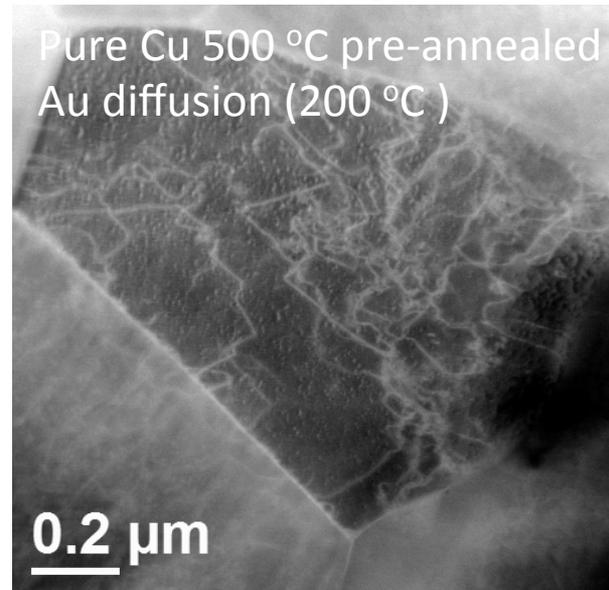
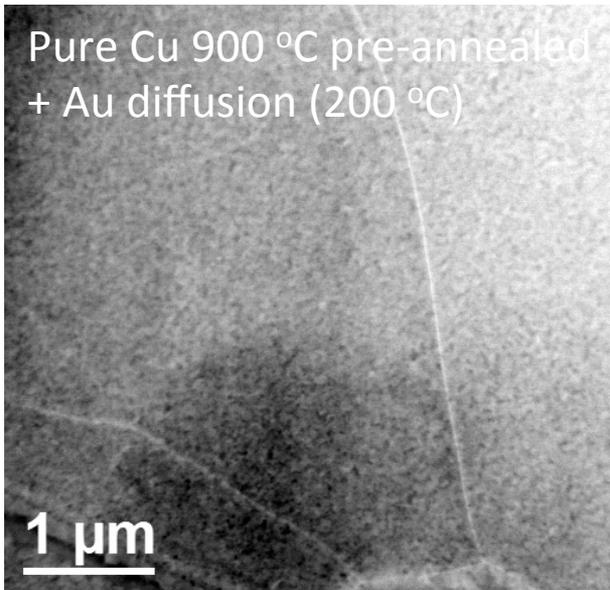


Cu of Varying Grain Size

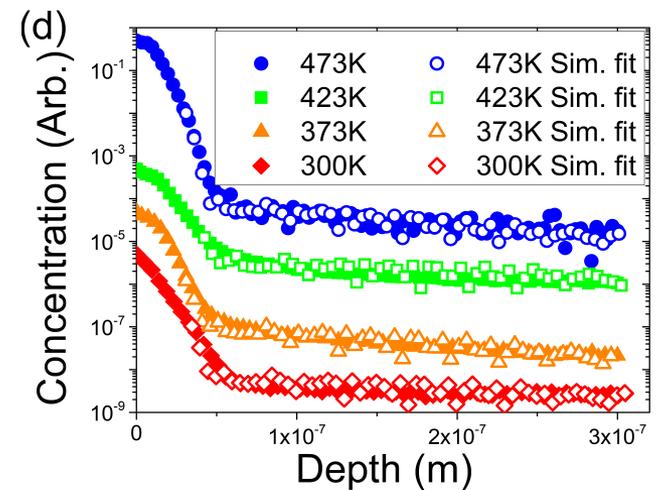
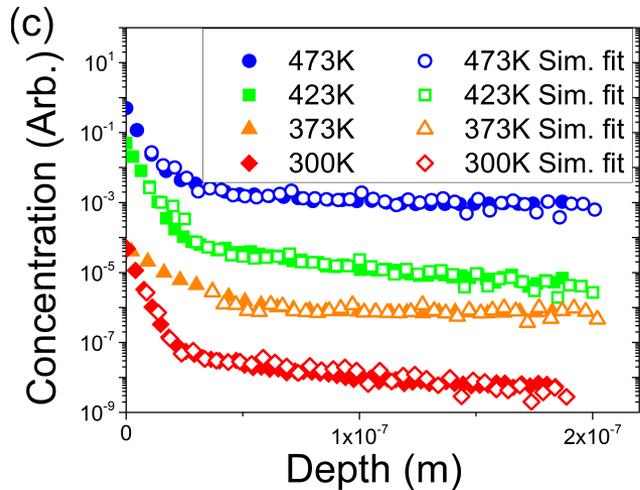
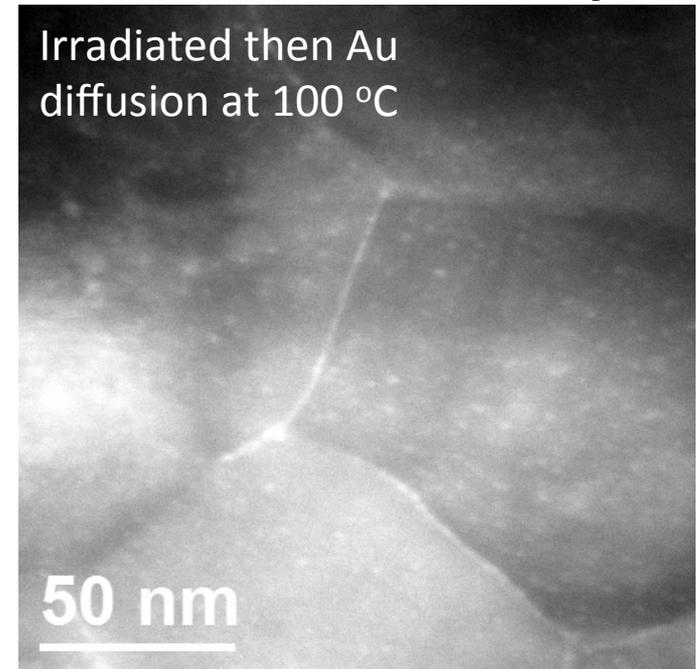
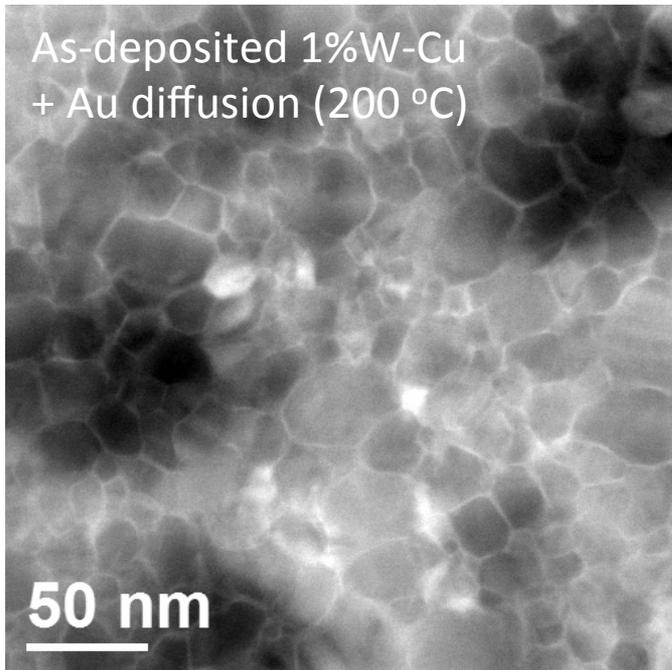


| Sample | Average Grain Size (nm) | | |
|-------------|-------------------------|-----------|----------|
| | X-ray | TEM | |
| | | w/o twins | w/ twins |
| Cu | 15.0 | ~150 | ~20 |
| Cu-W | 14.7 | ~30 | ~15 |
| Irrad. Cu-W | 17.0 | ~80 | ~25 |
| 773K Cu | 23.5 | ~2000 | ~280 |
| 1173K Cu | 47.0 | ~4000 | ~550 |

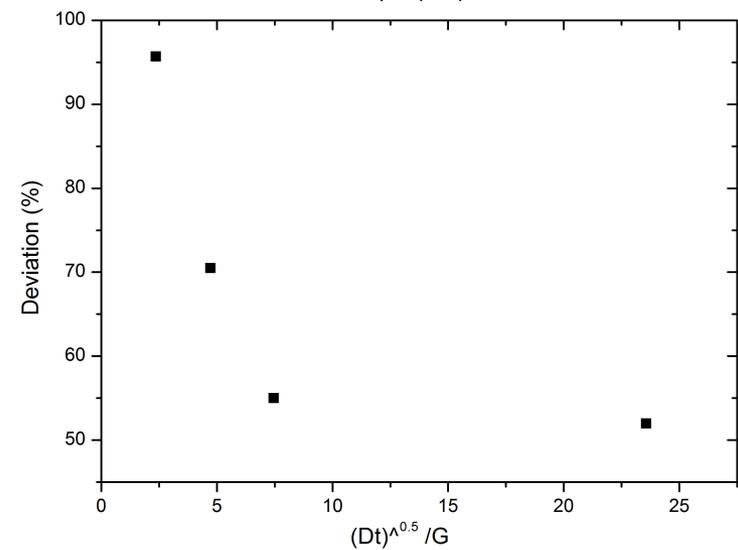
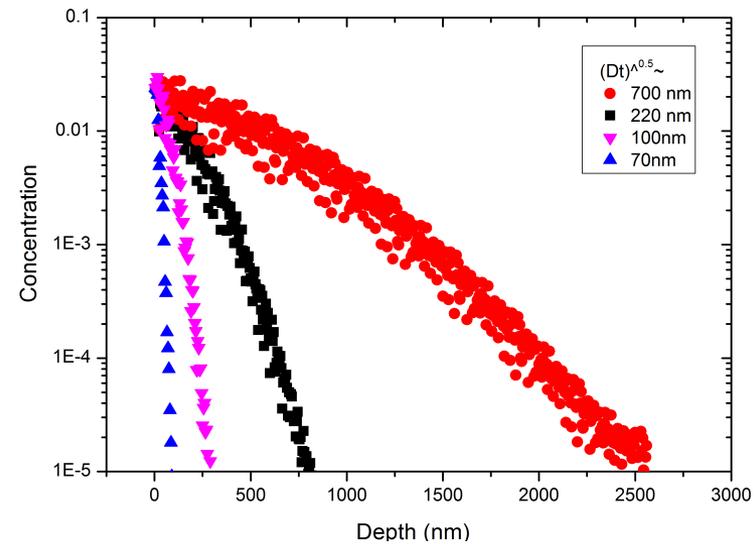
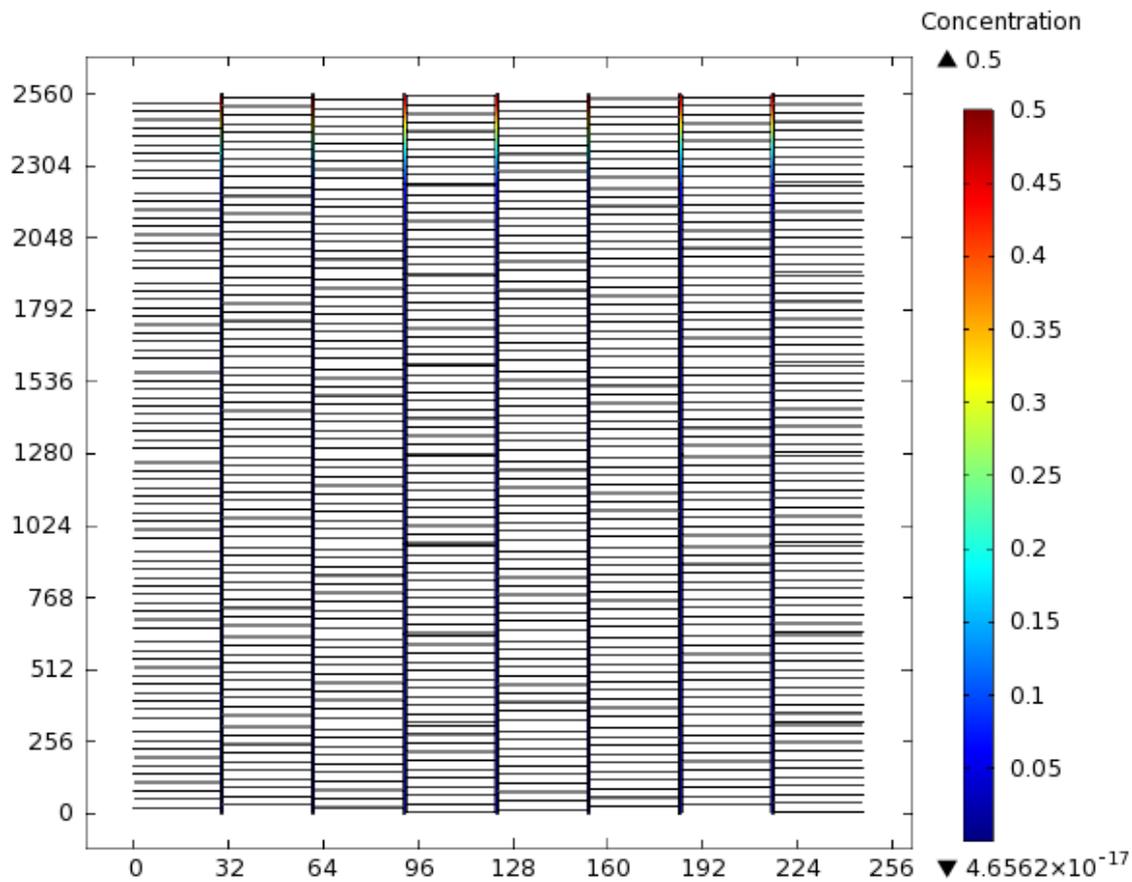
Au diffusion in Cu (Different G.S.'s)



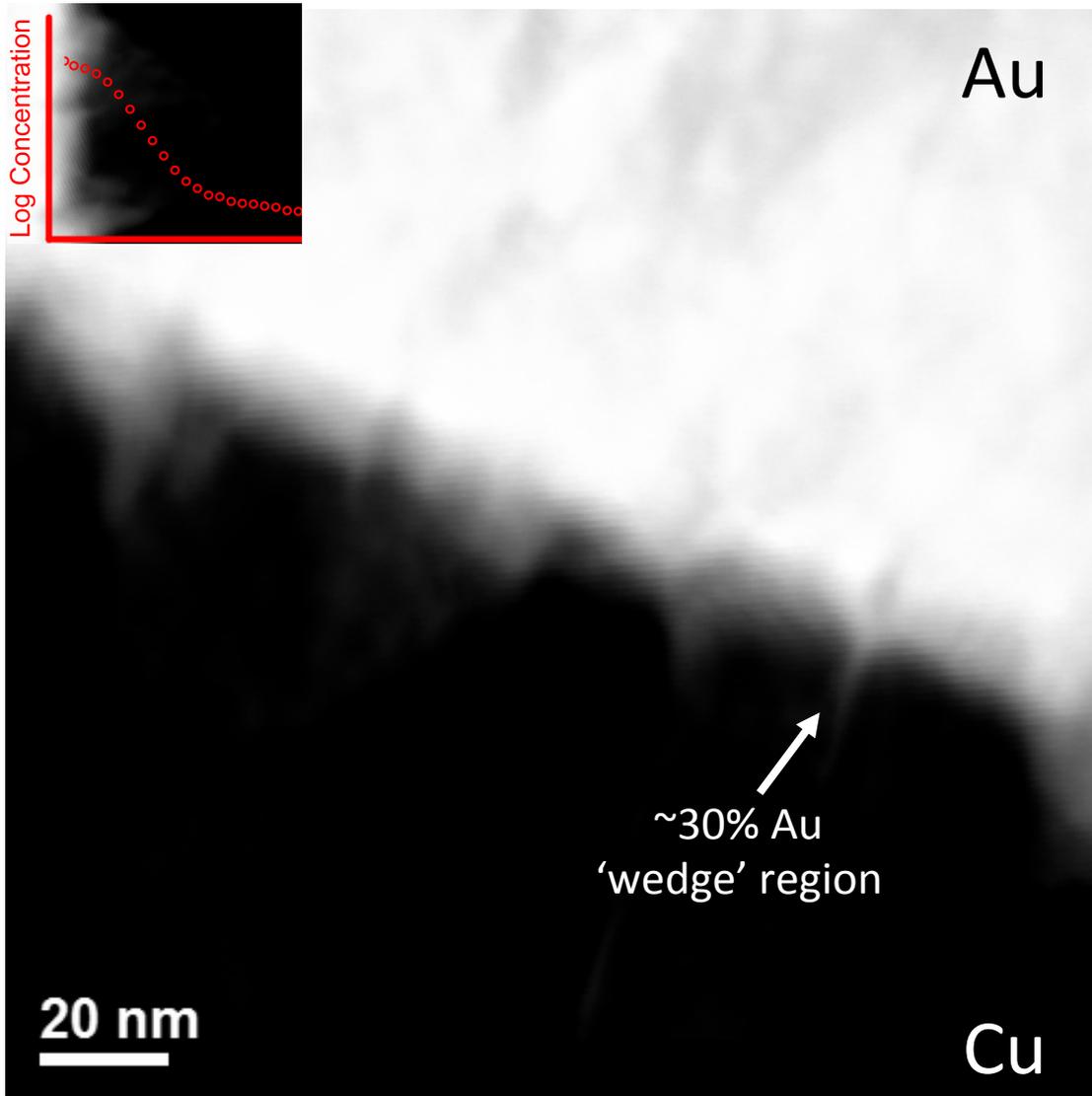
Au diffusion in Cu (Different G.S.'s)



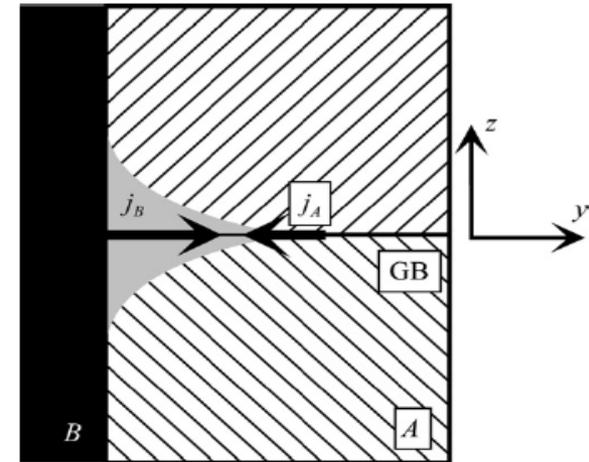
FE Simulation Nanograin Diffusion



Grain Boundary Kirkendall Effect



Kirkendall Effect induced
G.B. migration/lattice drift



A-B

Inequality Diffusion

$$j_A = -\frac{D_A n_A}{kT} \frac{\partial \mu_A}{\partial y}; \quad j_B = -\frac{D_B n_B}{kT} \frac{\partial \mu_B}{\partial y},$$

$$\frac{\partial C_B}{\partial t} = D_B \frac{\partial^2 C_B}{\partial y^2} + \frac{D_B C_B \Omega}{kT} \frac{\partial^2 \sigma}{\partial y^2} + \frac{\Omega D_B}{kT} \frac{\partial C_B}{\partial y} \frac{\partial \sigma}{\partial y}$$

Finite Element Simulation

Grain boundary diffusion in thin film model:

Near surface (10~20nm) D_{sb} different from D_b in film

