

Entropy and Enthalpy Effects on the Diffusivity of Different Grain Boundary Complexions

Shen J. Dillon

University of Illinois Urbana-Champaign



Acknowledgements

Students:

Kaiping Tai

Lin Feng

Salman Arshad

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N000141110678

NSF DMR#

1005813



Revolutionary Research . . . Relevant Results

Collaborators:

M. Harmer

G. Rohrer

J. Luo

J. Rickman

H. Chan

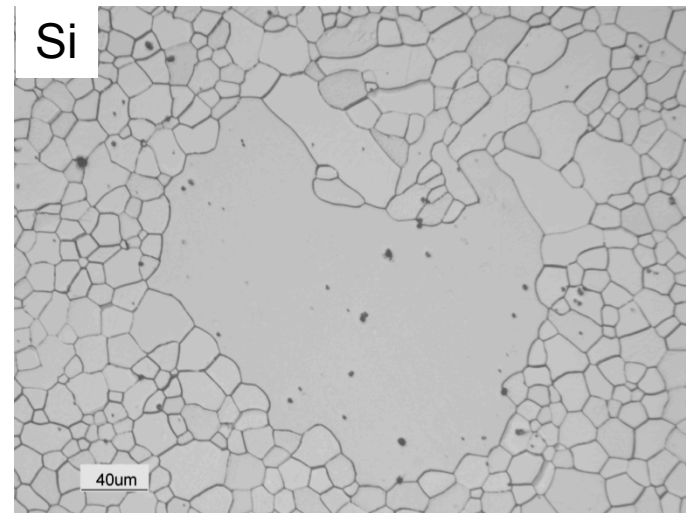
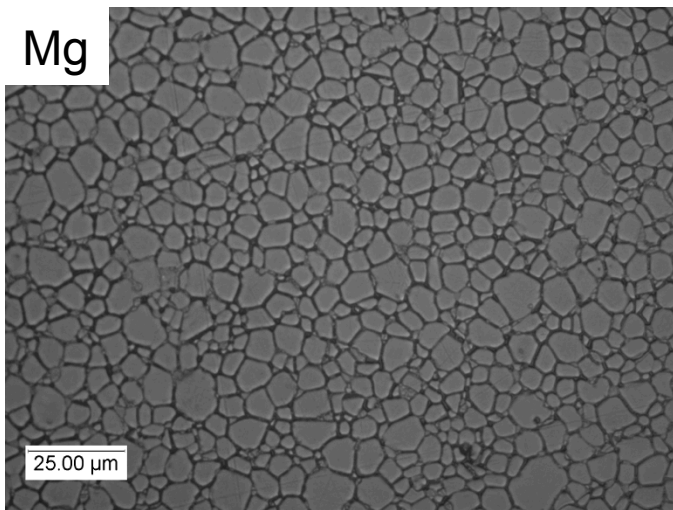
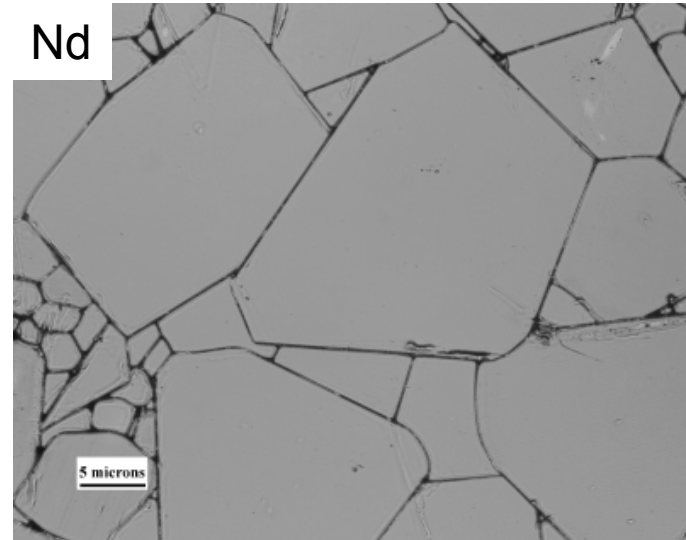
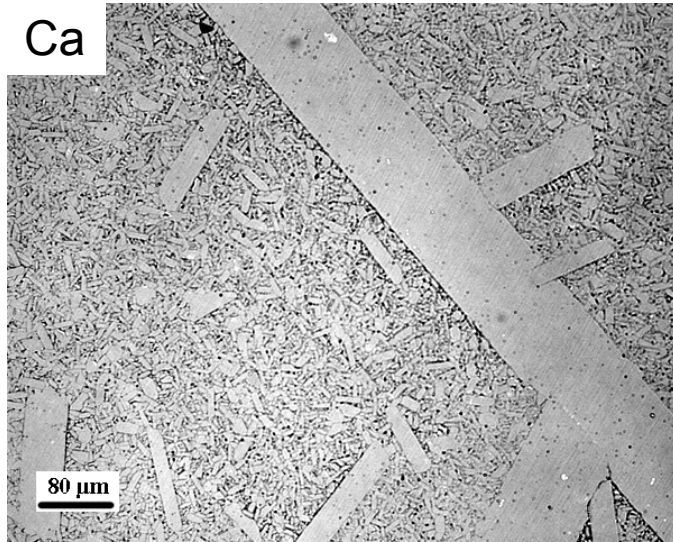
A. Rollett

M. Widom

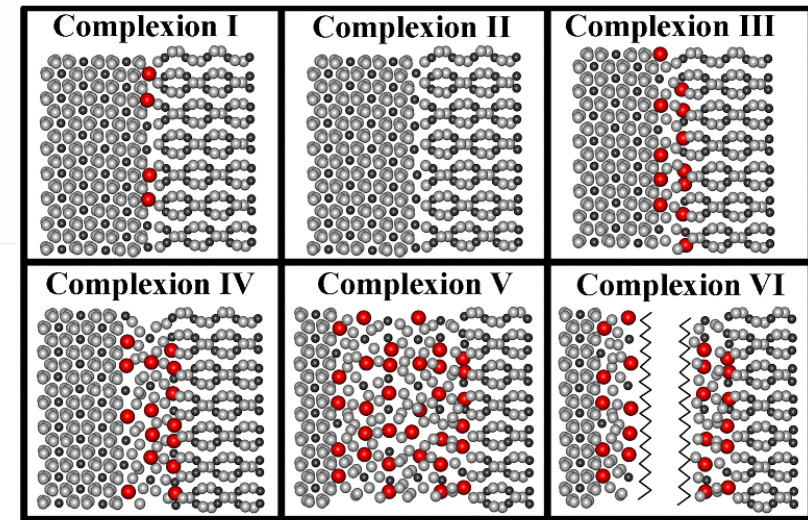
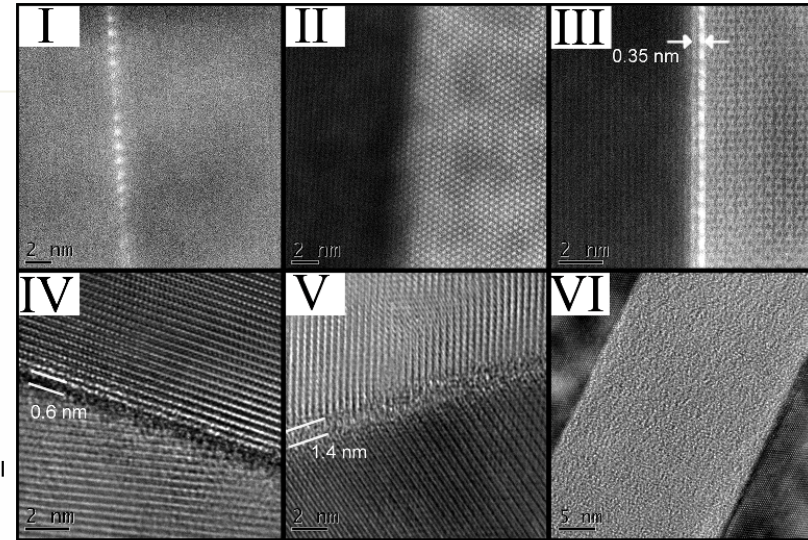
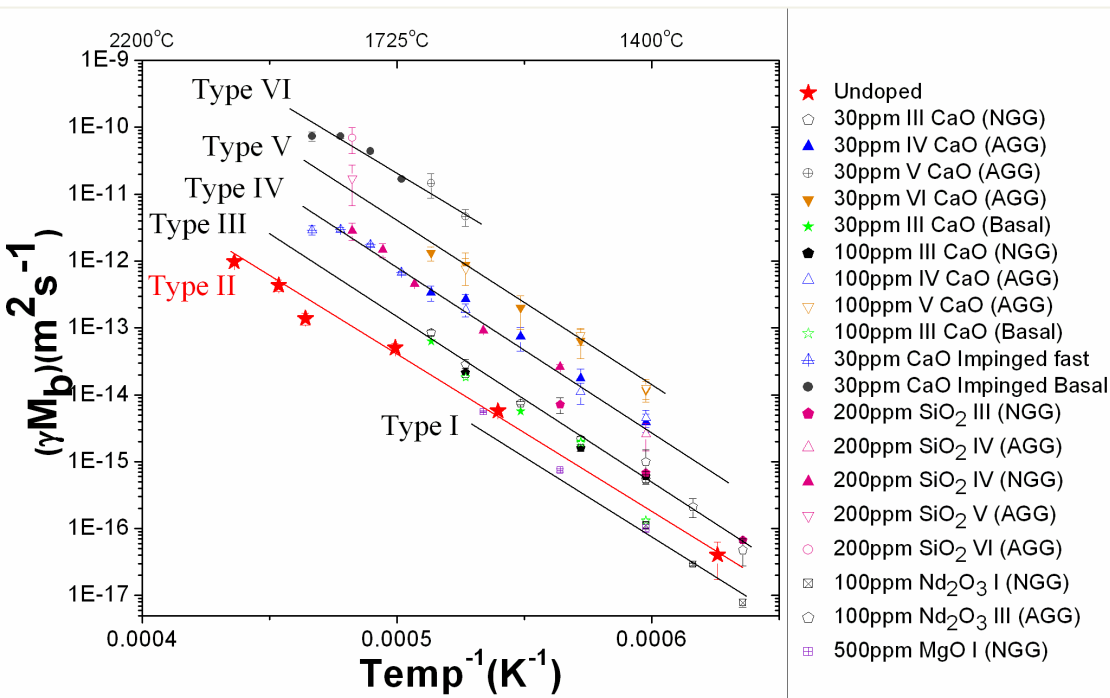
Overview

- Grain boundary complexions
- Approach to approximating S and H for G.B.
 - Measurements in Ni-Bi and Cu-Bi
- Diffusivity of nanograined samples
 - Is it different?
 - Does processing matter?

Grain Growth in Aluminum Oxide

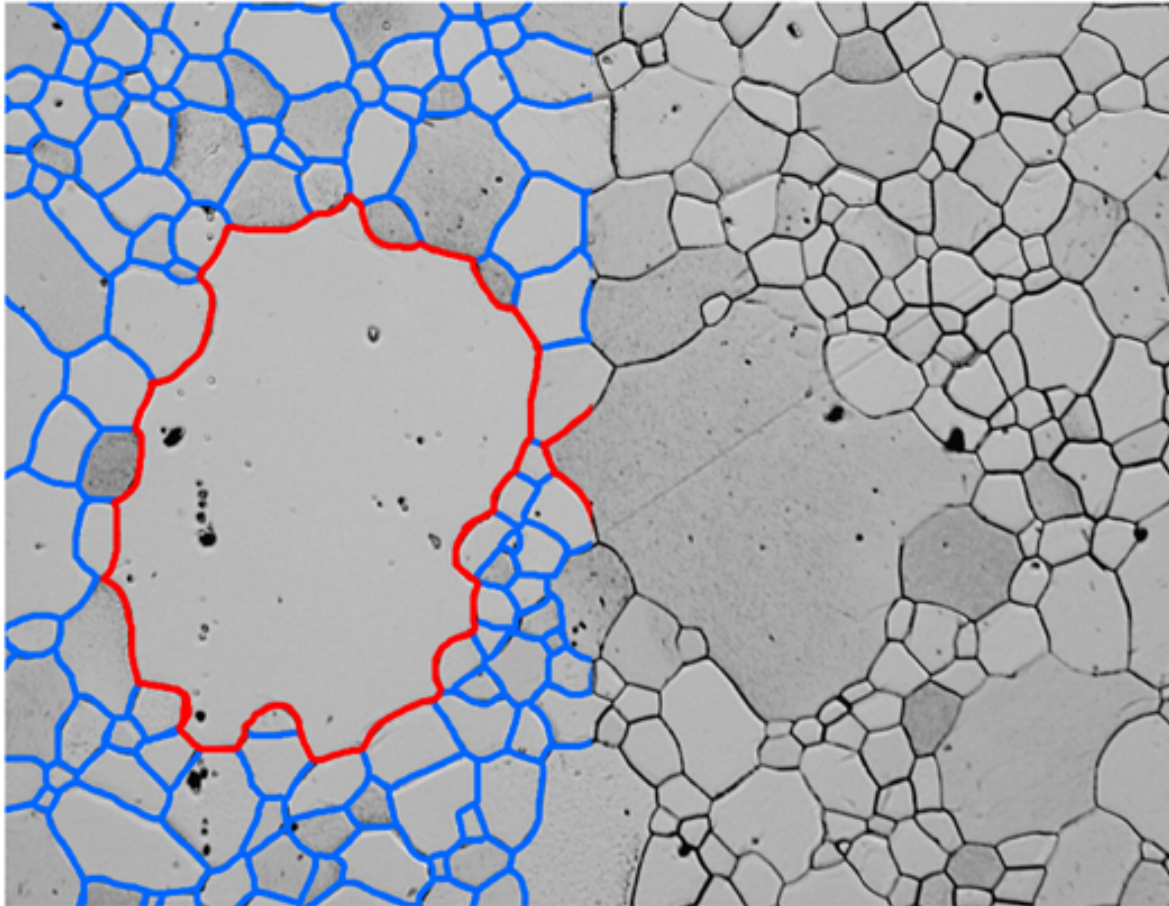


Grain Growth Kinetic Types



○ Oxygen Anion ● Dopant Cation ● Aluminum Cation

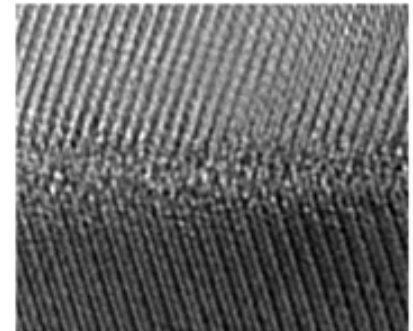
Simple Interpretation of AGG Mechanism



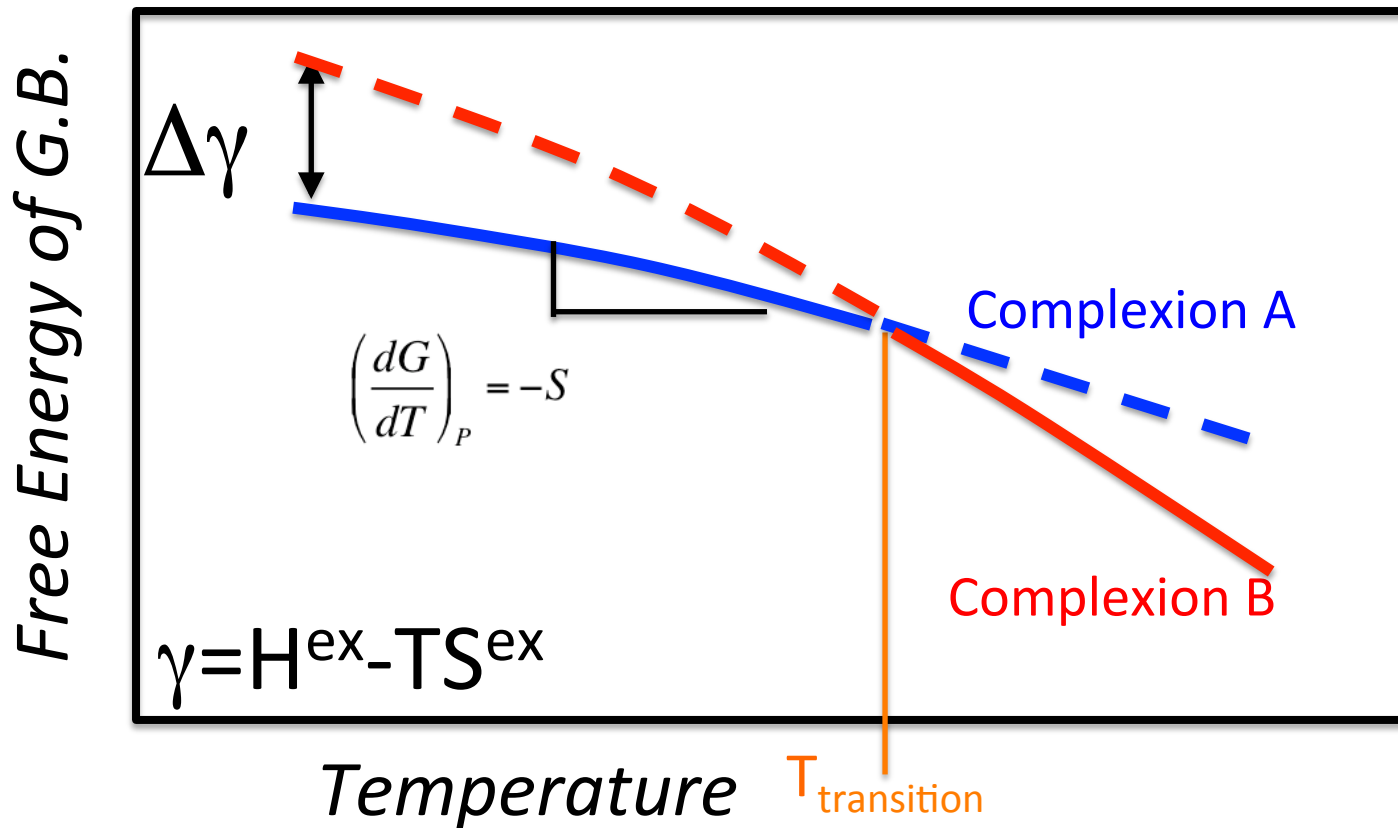
Complexion III



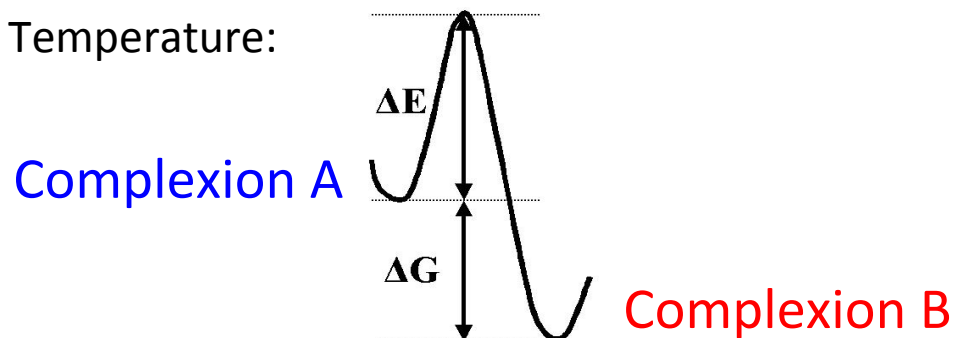
Complexion V



Simple Thermodynamic Model



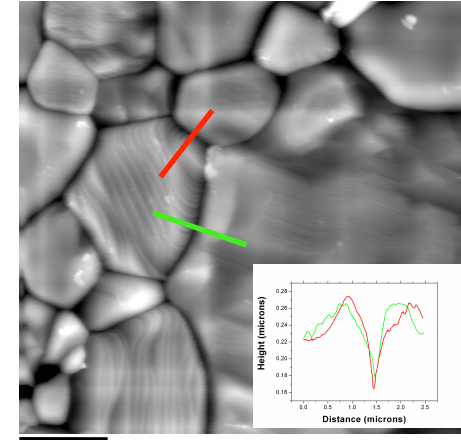
High Temperature:



Transition Driven by Energy Minimization

Table I. The Mean Relative Energies of Different Grain-Boundary Complexions Occurring as Normal and Abnormal Grains in Doped and Undoped Alumina Annealed at Different Temperatures

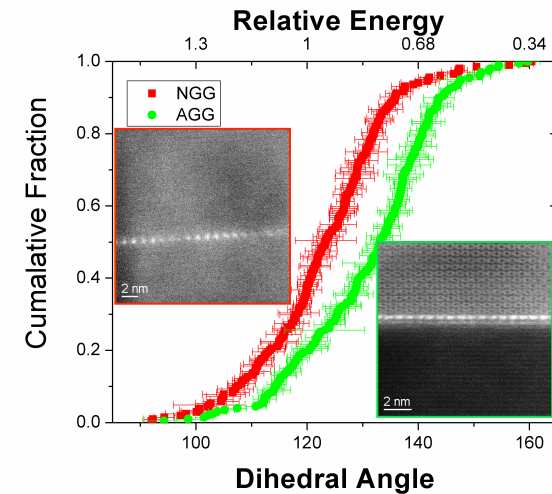
Chemistry	Temperature (°C)	Complexion	Relative energy	% energy change (complexion transition)
Undoped	1400	II (NGG)	1.11	
	2020	II (NGG)	1.08	
100 ppm-Nd ₂ O ₃	1400	I (NGG)	0.95	-16
	1400	III (AGG)	0.8	
100 ppm-Y ₂ O ₃	1400	I (NGG)	0.57	-46
	1400	III (AGG)	0.31	
500 ppm-MgO	1400	I (NGG)	1.07	-26
	1700	III (NGG)	0.79	
30 ppm-CaO	1200	I (NGG)	0.82	-20
	1200	III (AGG)	0.69	
200 ppm-SiO ₂	1200	I (NGG)	0.68	-10
	1200	III (AGG)	0.61	
30 ppm-CaO	1400	III (NGG)	1.02	0.1
	1400	IV+ (AGG)	1.02	
	1400	III (Basal plane)	0.77	
	1400	III (NGG)	0.65	9.5
200 ppm-SiO ₂	1400	IV (AGG)	0.71	
	1750	IV (NGG)	0.98	-1.7
200 ppm-SiO ₂	1750	V+ (AGG)	0.96	



2 μm

Relative Contributions?

$$\gamma = H^{\text{ex}} - TS^{\text{ex}}$$



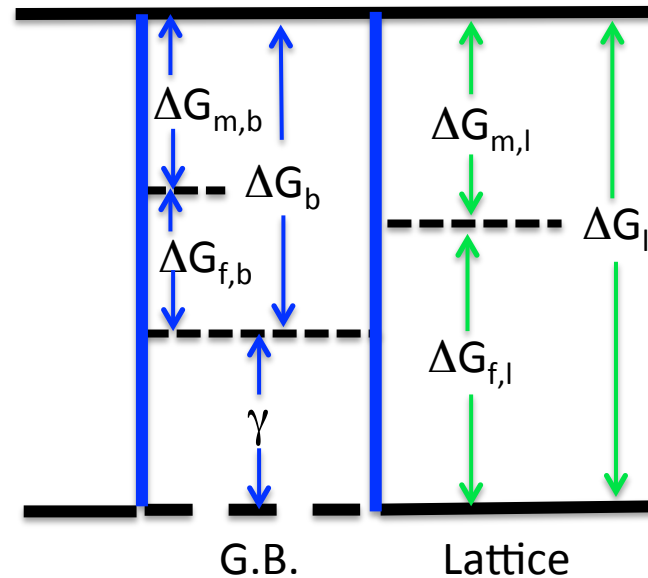
Borisov Model: Linking Thermos & Kinetics

$$D = ga^2 C_v \Gamma$$

$$C_v = \exp(\Delta S_f/k) \exp(-\Delta H_f/kT)$$

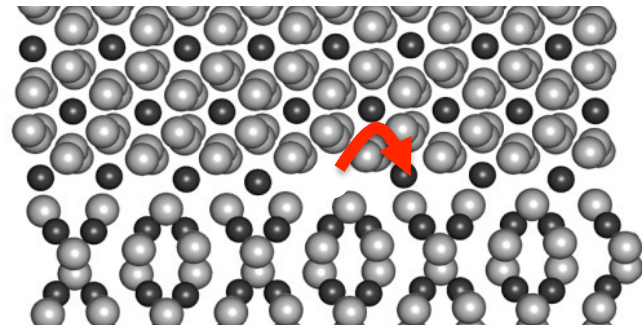
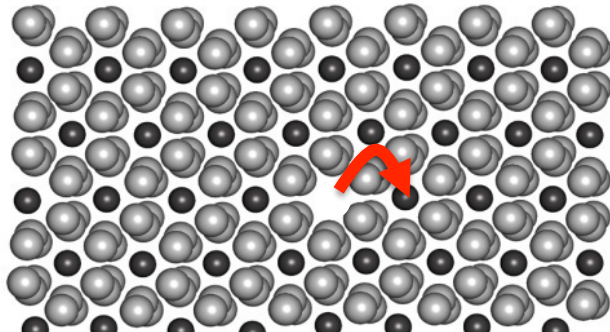
$$\Gamma = v^* \exp(-\Delta H_m/kT)$$

Assumption: $D_b = ga_o^2 v^* \exp[-(\Delta G_l - \rho \gamma')/kT]$ Remember $\rho \gamma' = \Delta \bar{G}_b - \Delta \bar{G}_l$

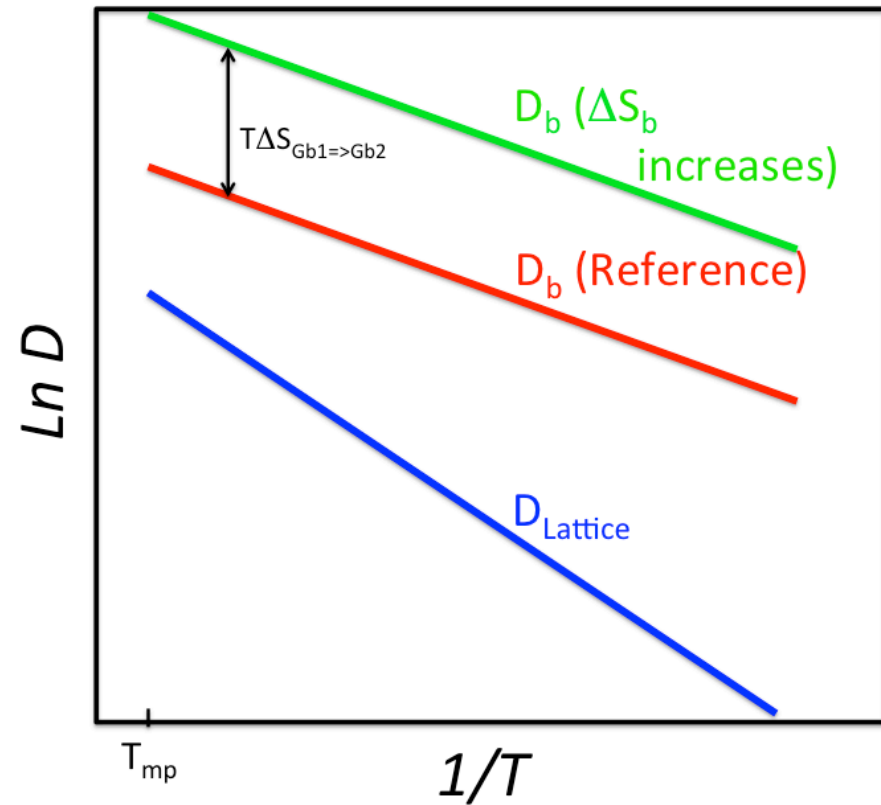
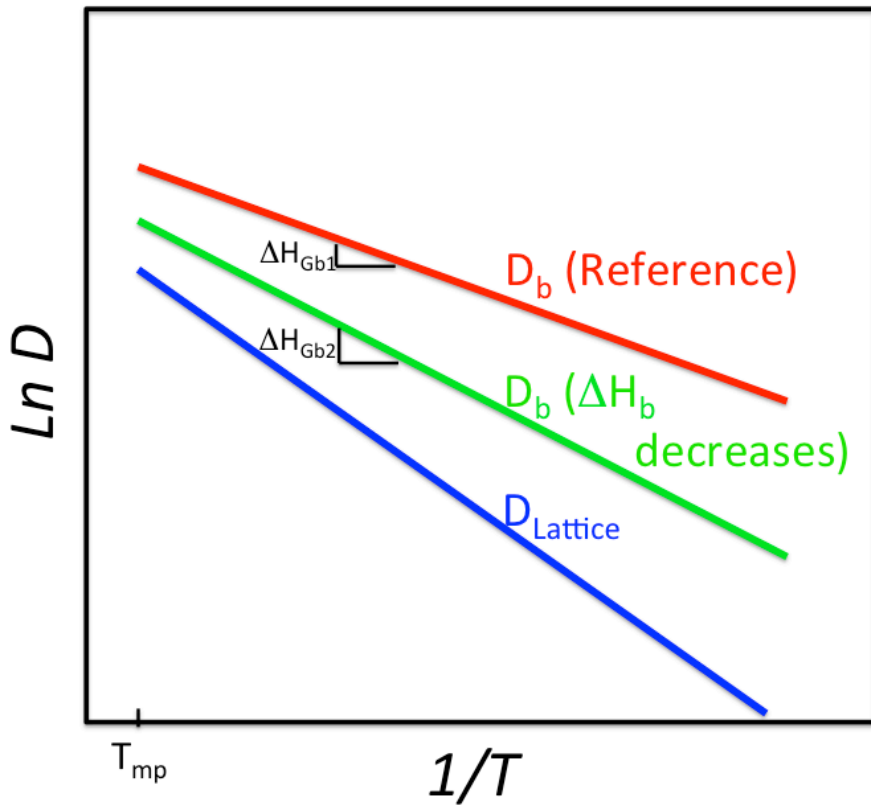


Adapted from Gupta *Met Trans.* (1977)

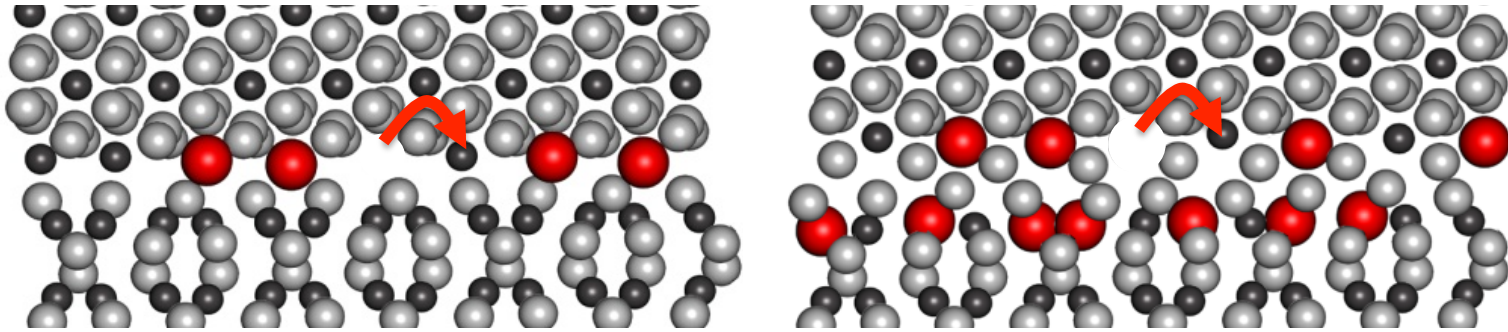
$$\gamma' = \rho^{-1} \left[kT \ln \left(\frac{ga_b^2 v_b^*}{ga_l^2 v_l^*} \right) - T(\Delta S_{f,b} - \Delta S_{f,l}) + (\Delta H_{f,b} - \Delta H_{f,l}) + (\Delta H_{m,b} - \Delta H_{m,l}) \right] = \rho^{-1} \left[kT \ln \left(\frac{D_b}{D_l} \right) \right]$$



General Predictions of Borisov Model

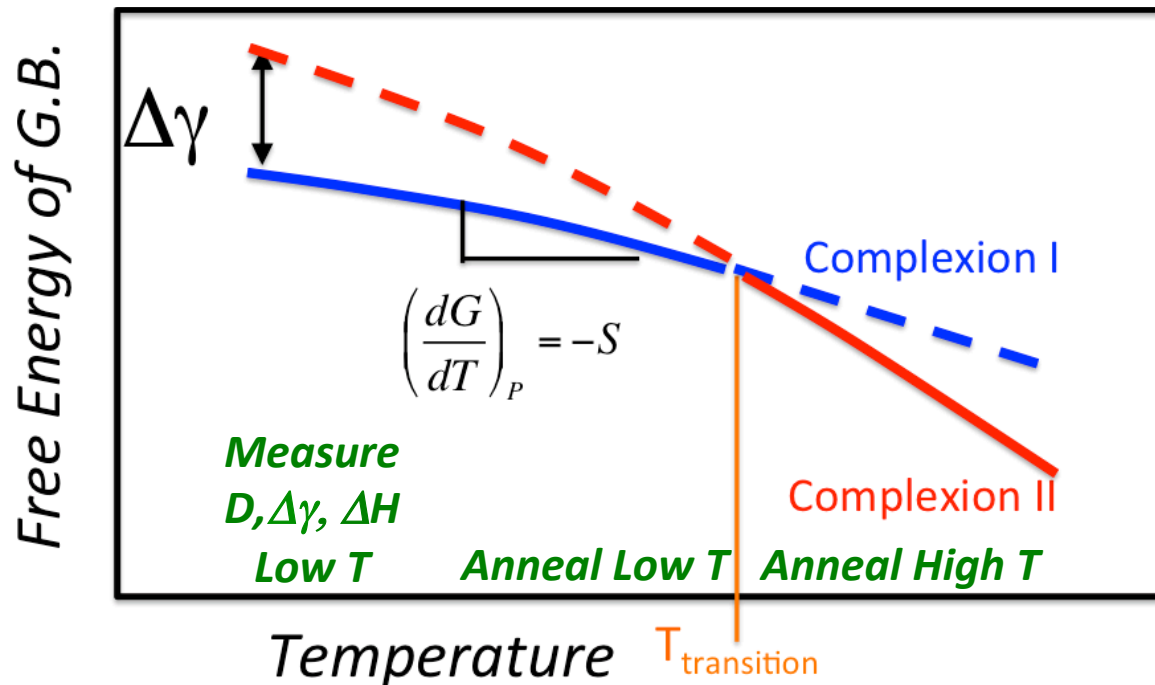


Comparing Complexions (ΔH & ΔS)

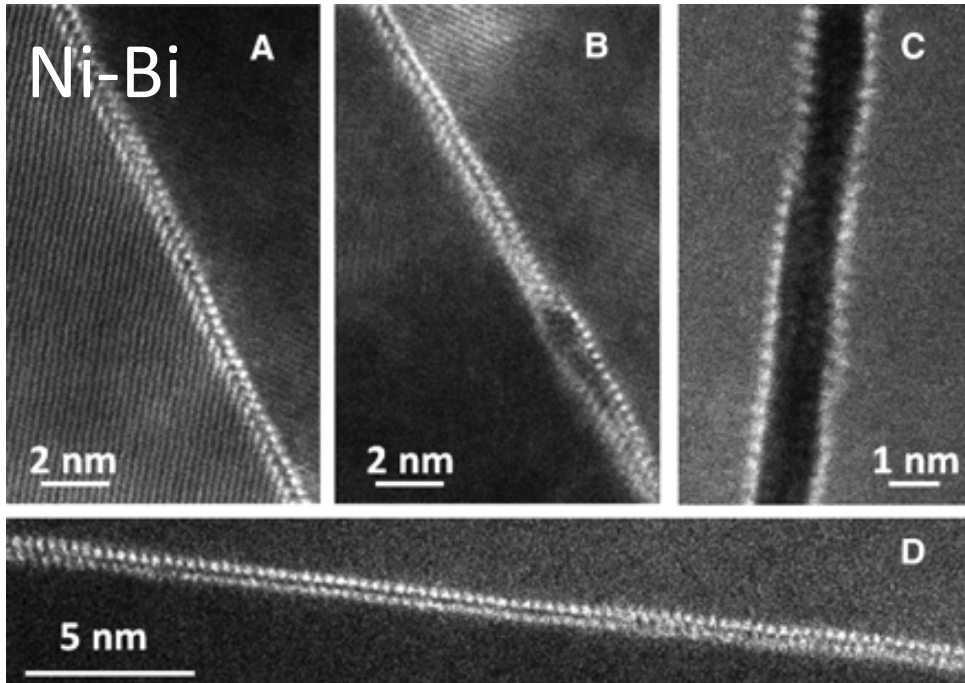


$$\Delta\gamma' = \rho^{-1} \left[kT \ln \left(\frac{g a_{II}^2 \nu_{II}^*}{g a_I^2 \nu_I^*} \right) - T(\Delta S_{f,I} - \Delta S_{f,II}) + (\Delta H_{f,I} - \Delta H_{f,II}) + (\Delta H_{m,I} - \Delta H_{m,II}) \right] = \rho^{-1} \left[kT \ln \left(\frac{D_I}{D_{II}} \right) \right]$$

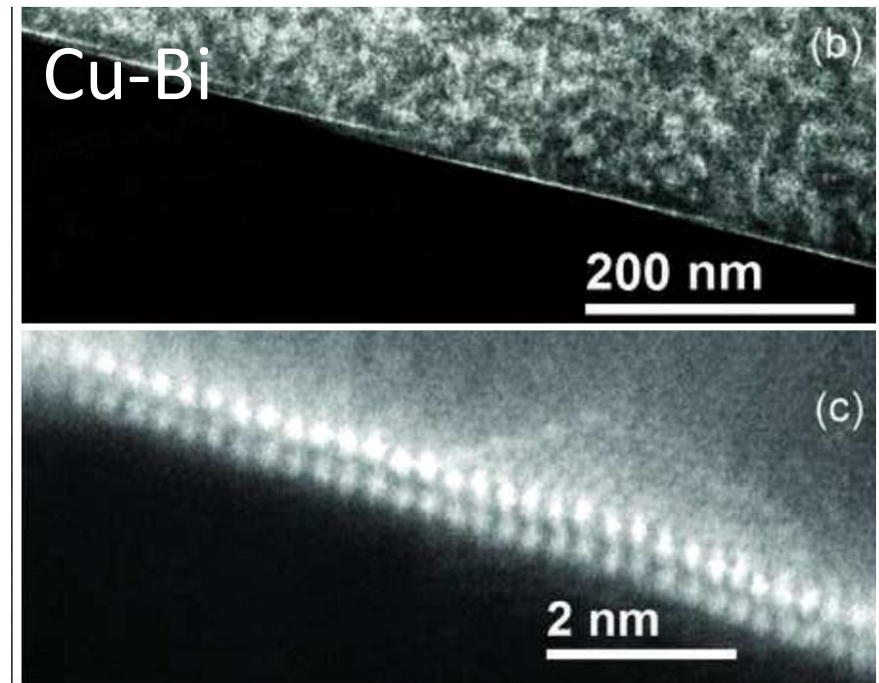
Approach:



2 Simple Model Systems

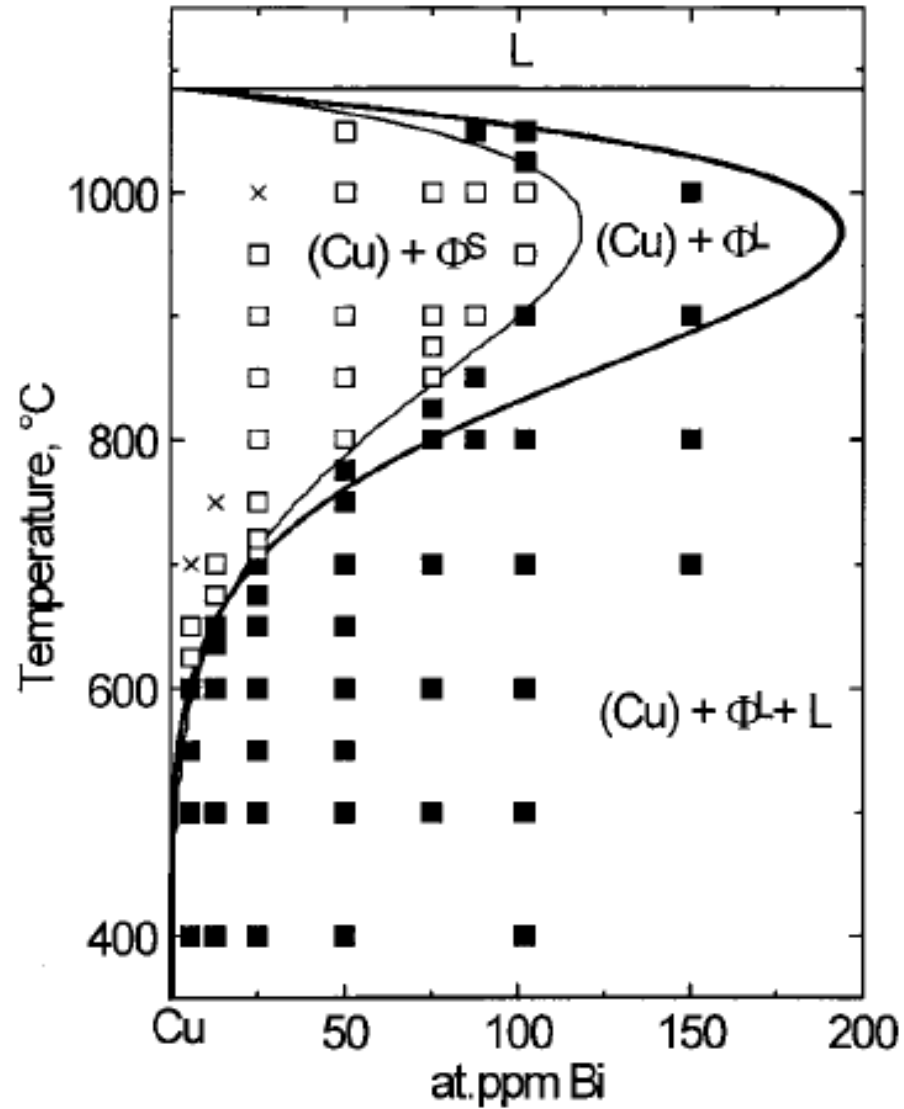


Luo et al. *Science* (2011)

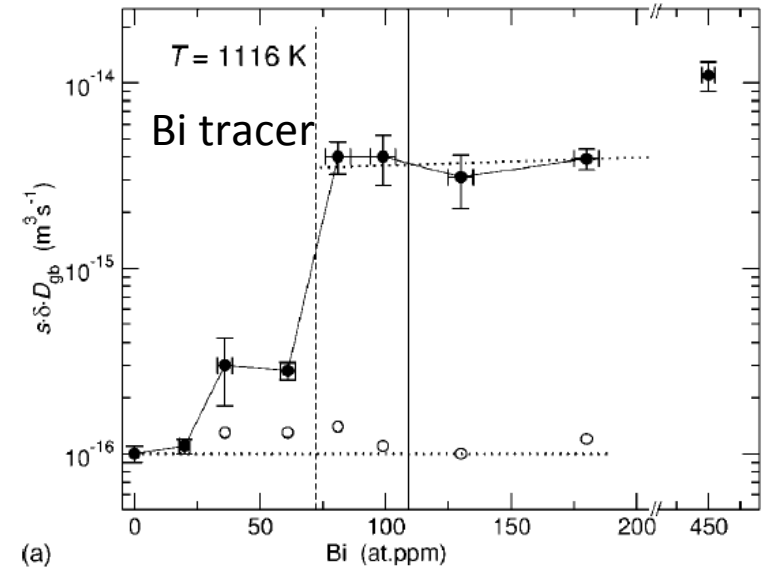
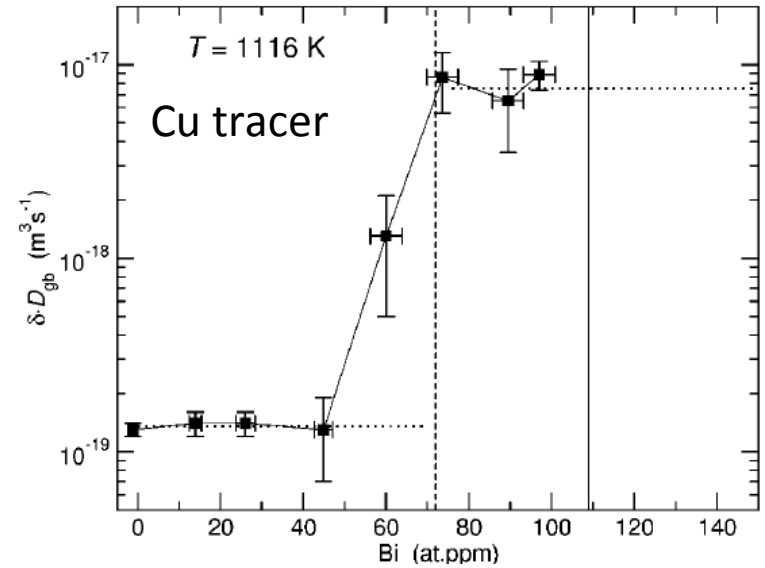


Kundu et al. *Scripta Mater.* (2012)

Prior Work on Cu-Bi



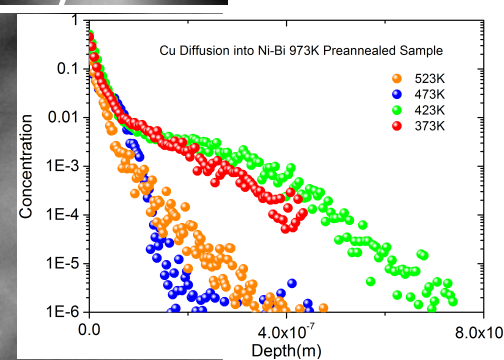
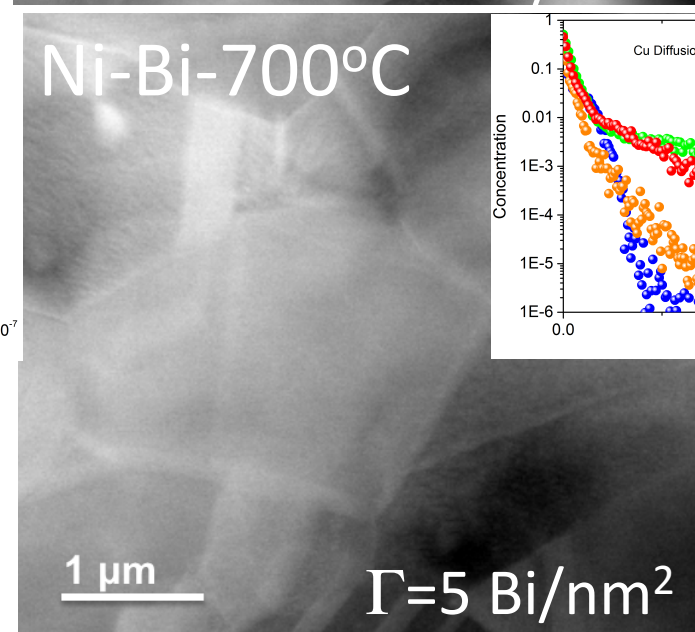
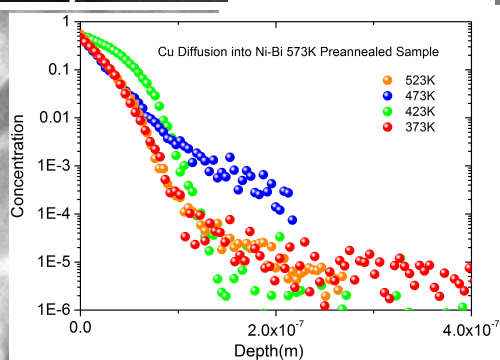
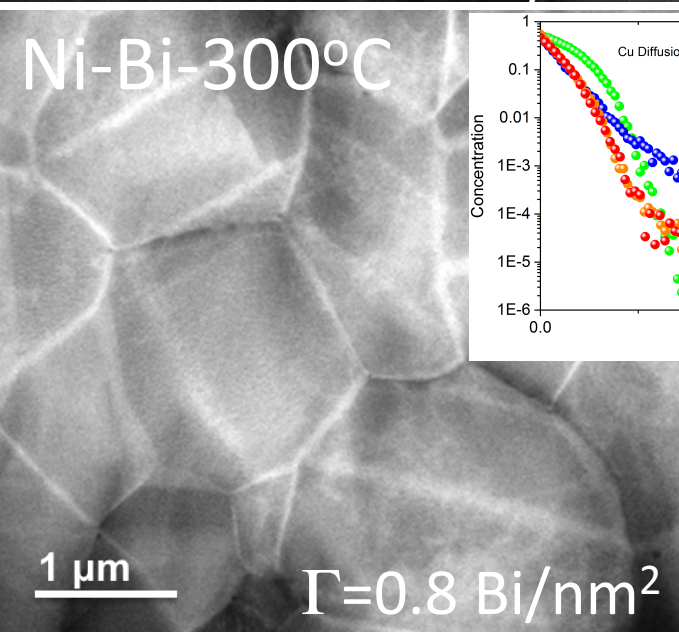
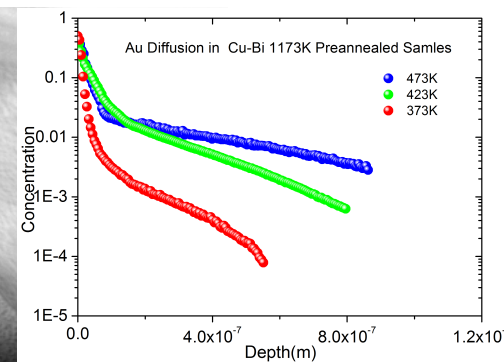
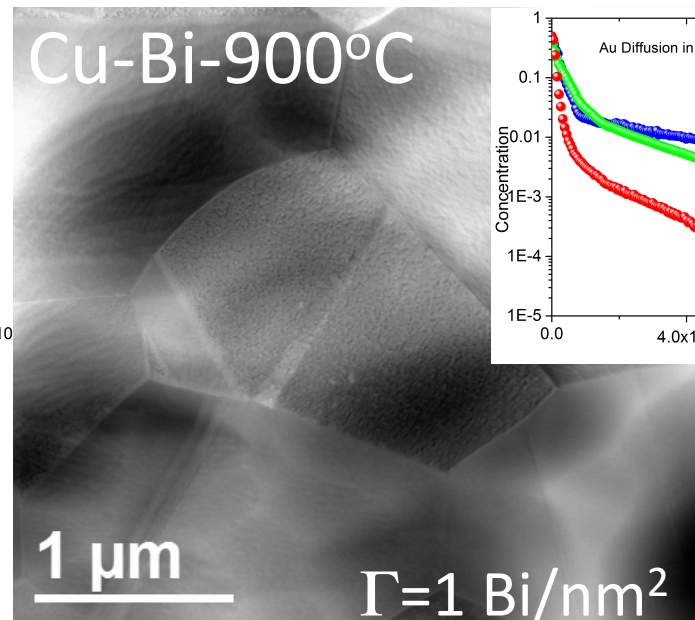
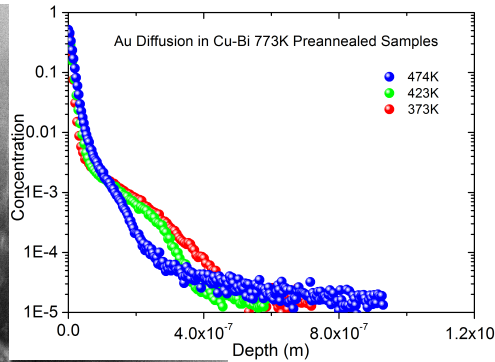
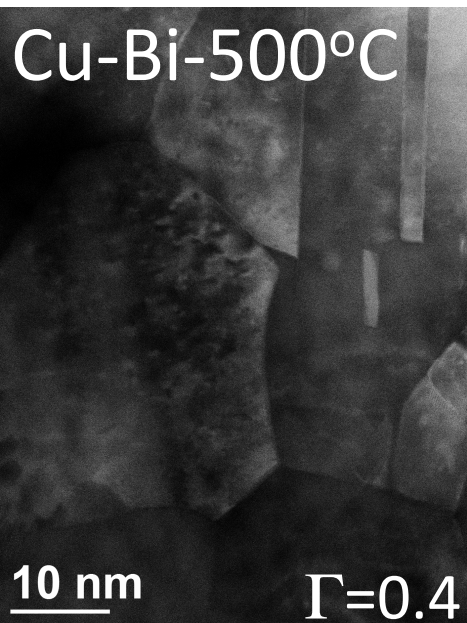
Chang et al. *Acta Mater.* (1999)



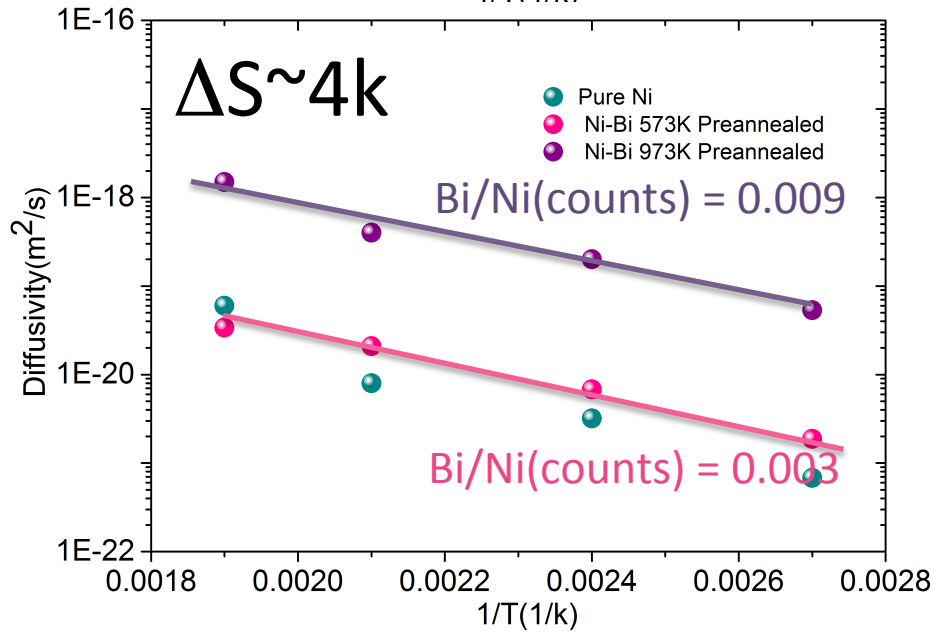
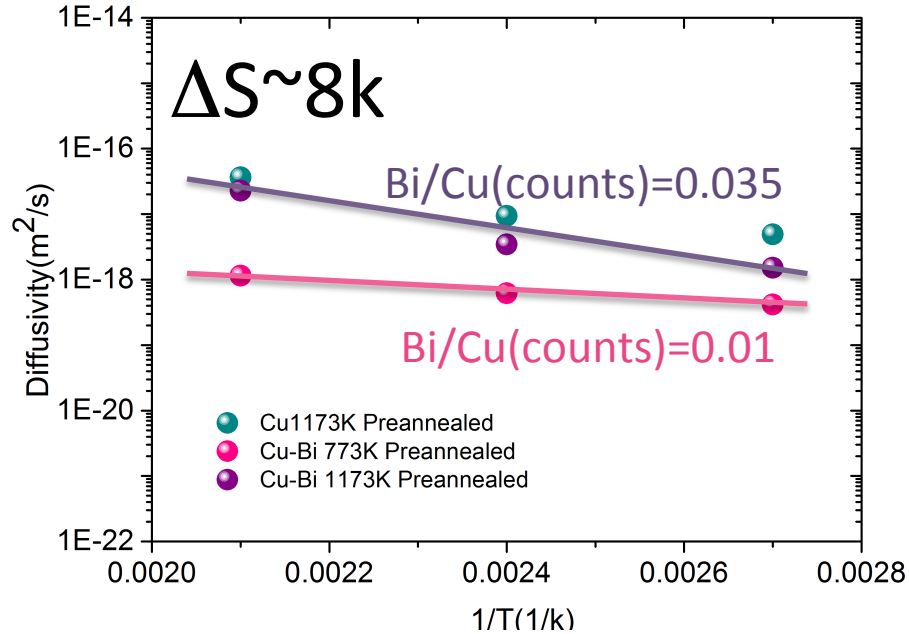
(a)

Divinski et al. *PRB* (2005)

Chemical Diffusion into Bi doped Ni/Cu

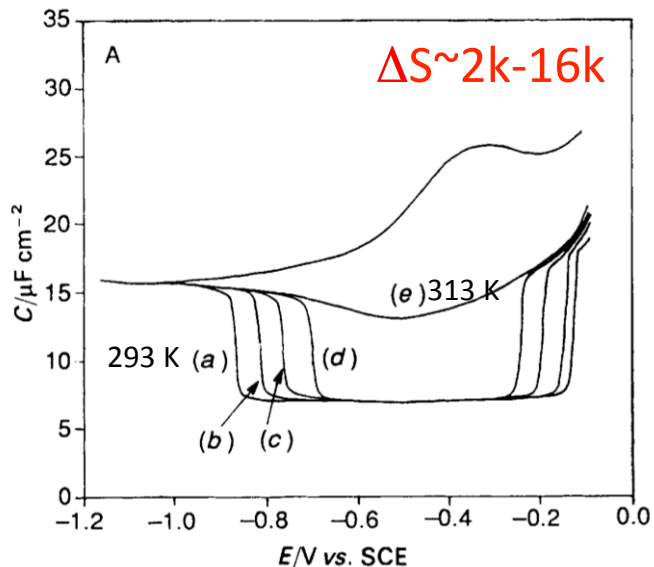


Entropy Difference between Complexions



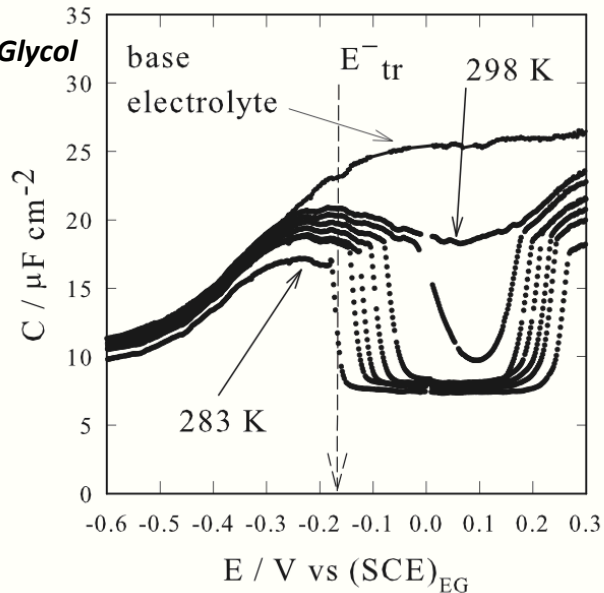
Comparison w/ Related Measurements

Adenine
Hg/Aqueous

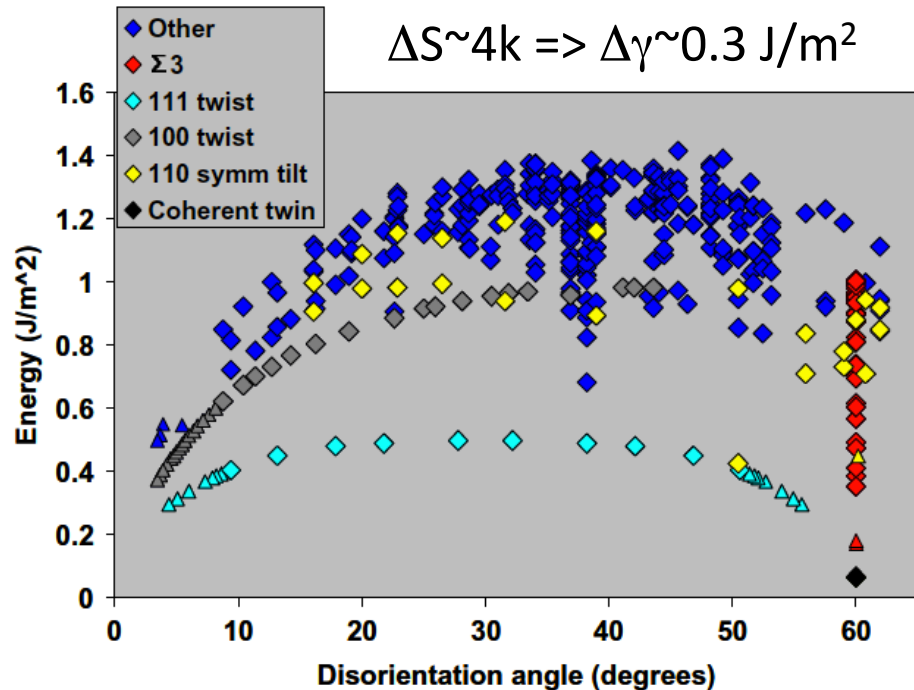


Fontanesi *J. Chem. Soc. Faraday Trans.* (1994)

Anthracene
Hg/Ethylene Glycol



Fontanesi *Entropy* (2010)

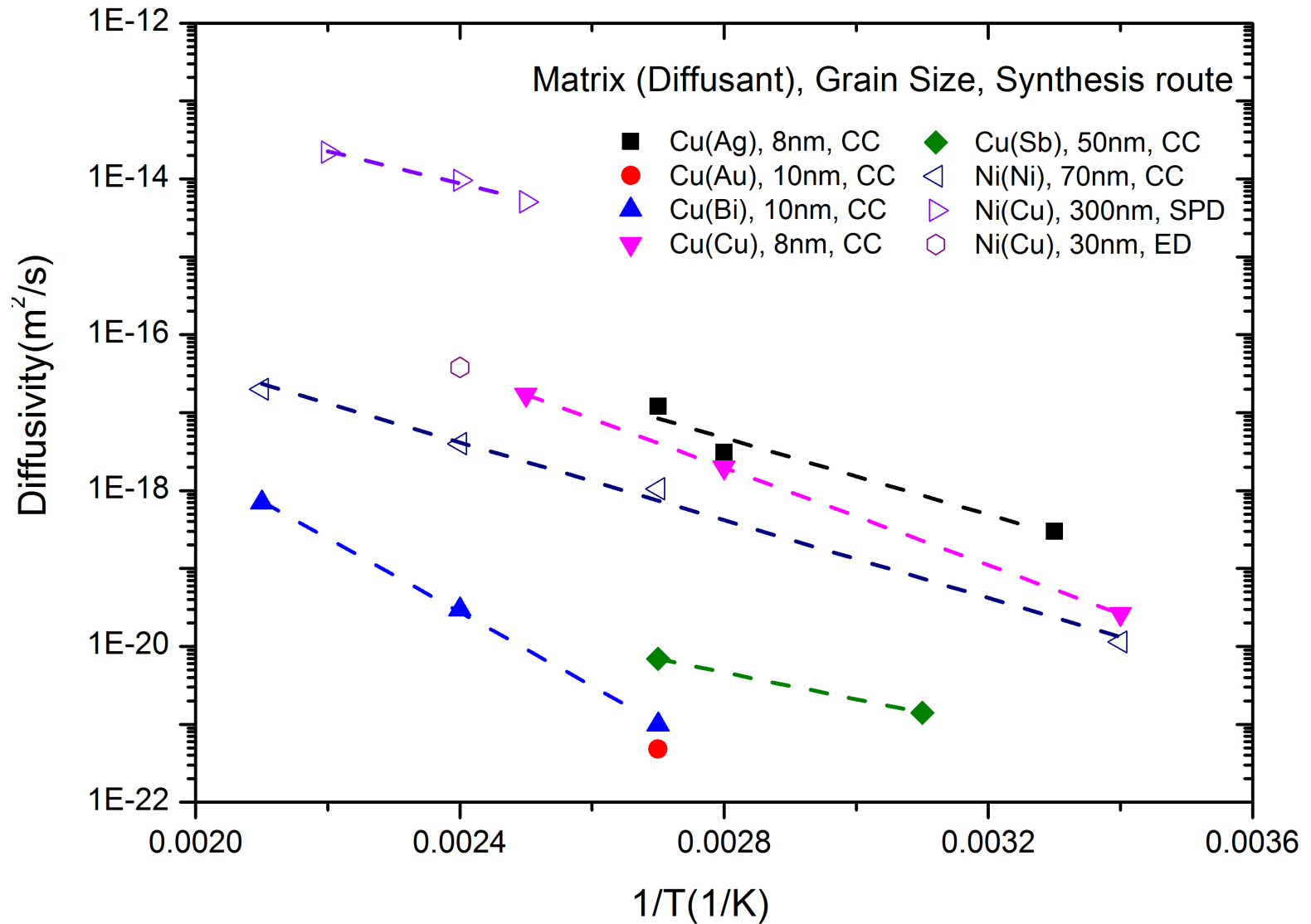


Olmstead et al. *Acta Mater.* (2009)

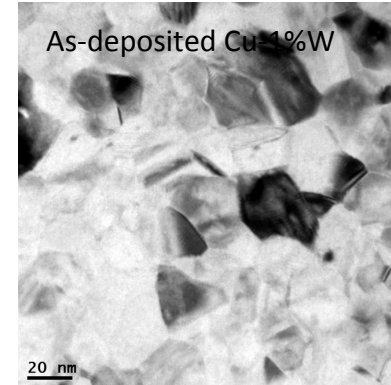
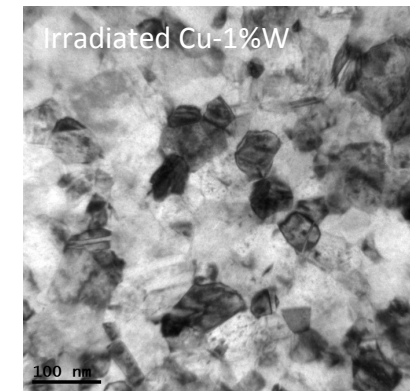
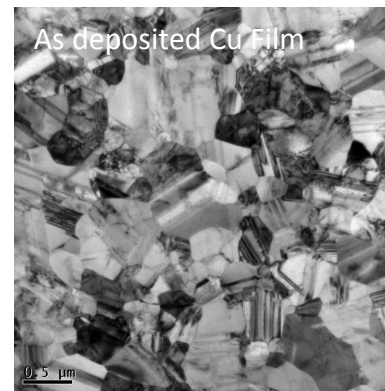
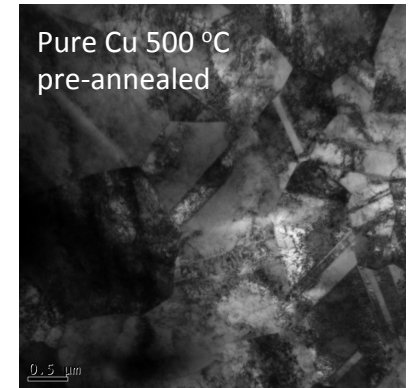
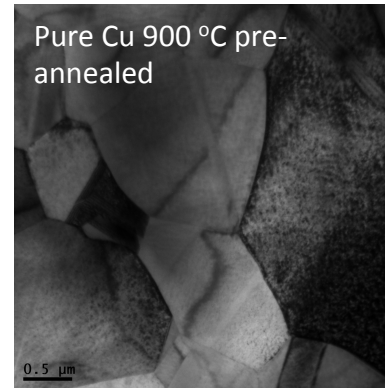
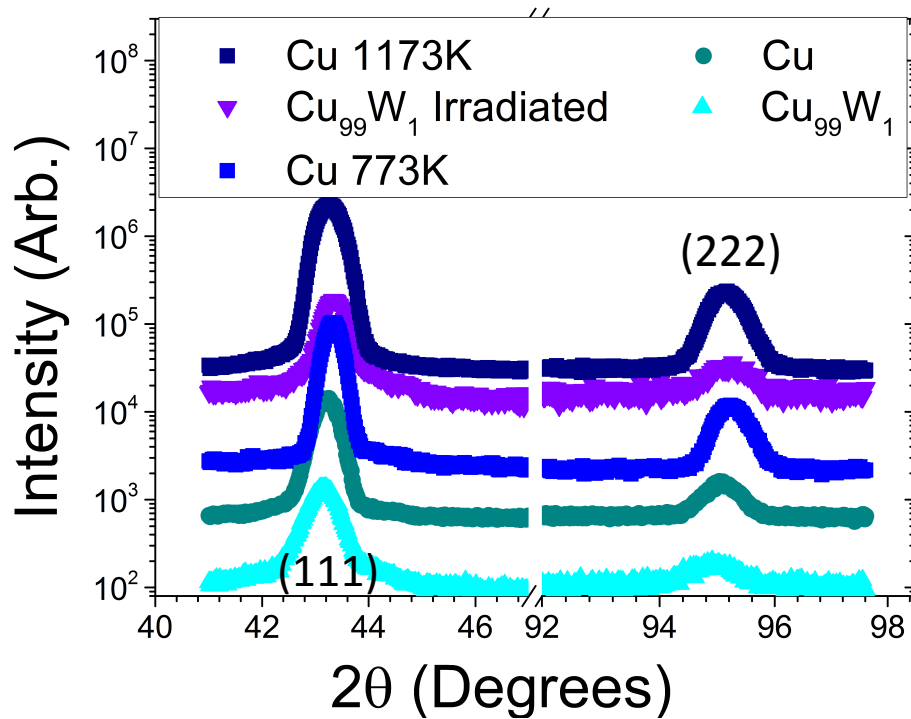
Chemistry	Complexion	Relative energy	% energy change (complexion transition)
Undoped	II (NGG)	1.11	
	II (NGG)	1.08	
100 ppm-Nd ₂ O ₃	I (NGG)	0.95	-16
	III (AGG)	0.8	
100 ppm-Y ₂ O ₃	I (NGG)	0.57	-46
	III (AGG)	0.31	
500 ppm-MgO	I (NGG)	1.07	-26
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	III (AGG)	0.69	
200 ppm-SiO ₂	I (NGG)	0.68	-10
	III (AGG)	0.61	

Dillon et al. *JACerS* 2010

GB Diffusion in Nanograin Alloys- Motivation

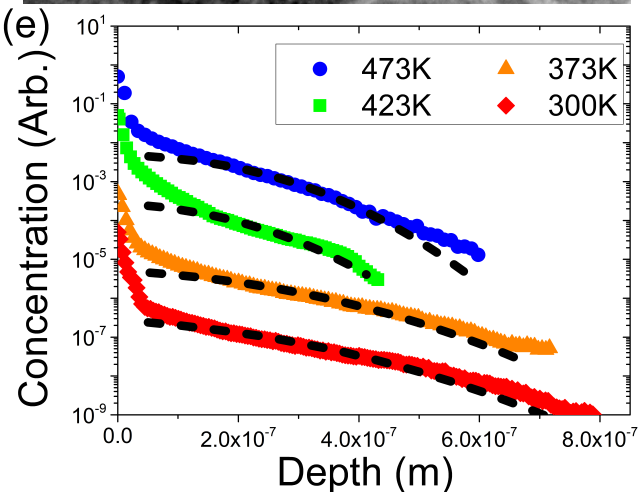
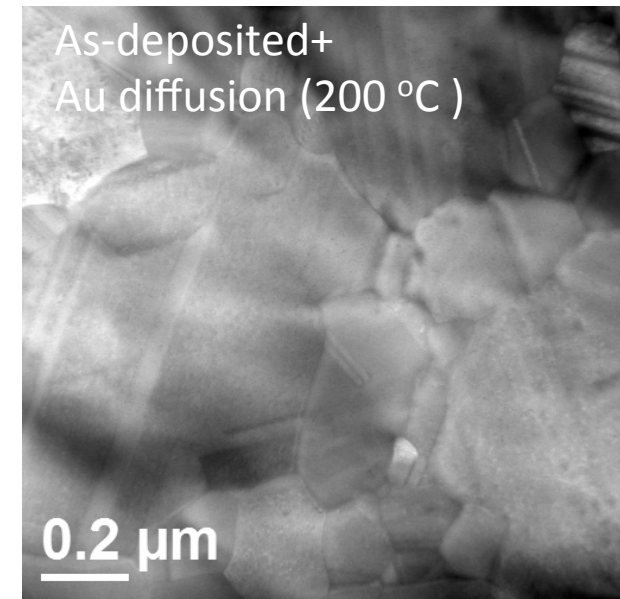
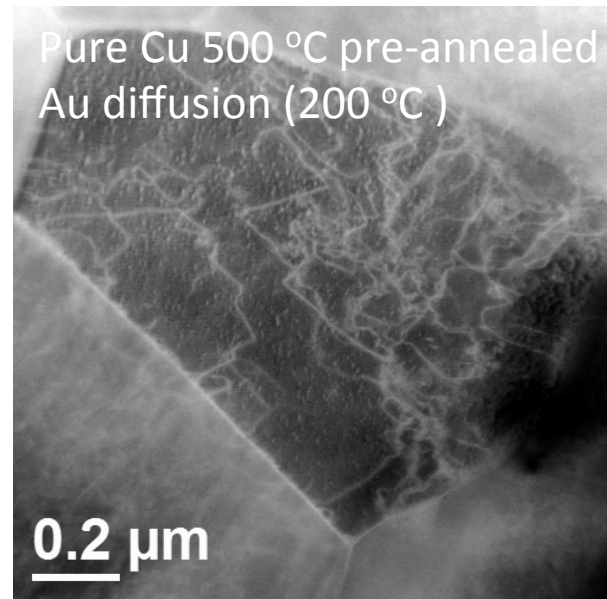
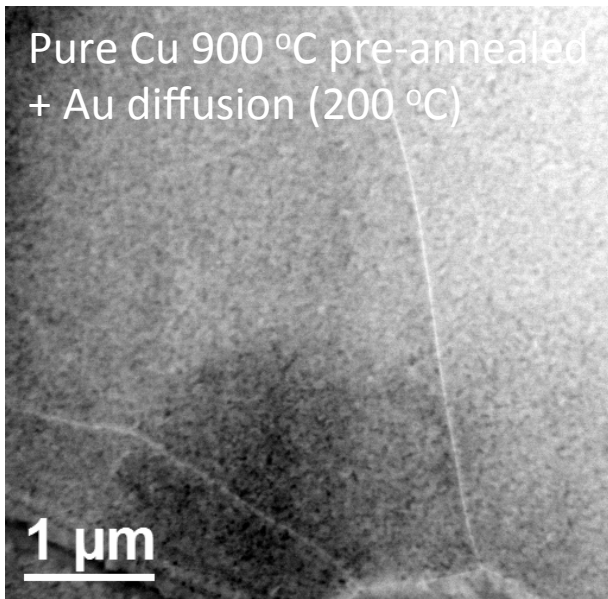


Cu Thin Films of Varying Grain Size

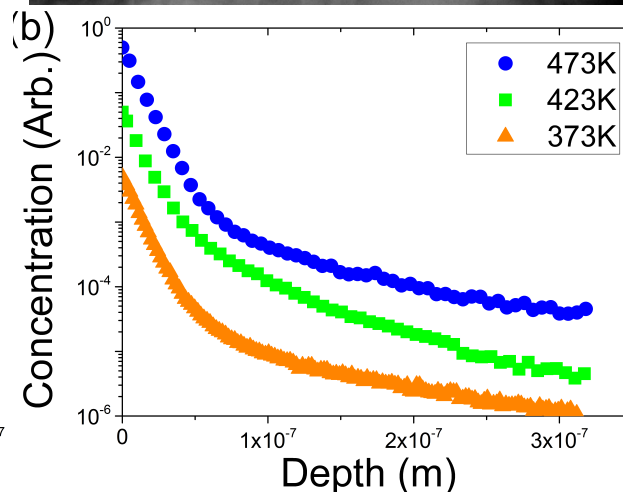


Sample	Average Grain Size (nm)		
	X-ray	TEM	
		w/o twins	w/ twins
Cu	15.0	~150	~20
Cu-W	14.7	~30	~15
Irrad. Cu-W	17.0	~80	~25
773K Cu	23.5	~2000	~280
1173K Cu	47.0	~4000	~550

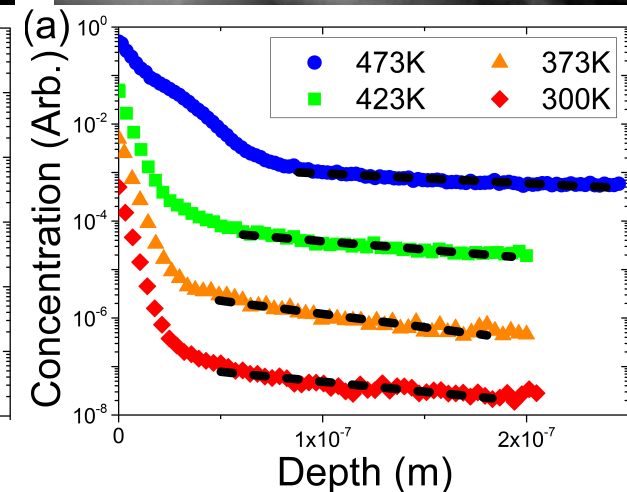
Au diffusion in Cu (Different G.S.'s)



$$\bar{c}(z) \approx \bar{c}_{gb}(z) \propto \operatorname{erfc}\left(\frac{z}{2\sqrt{D_{gb}t}}\right)$$

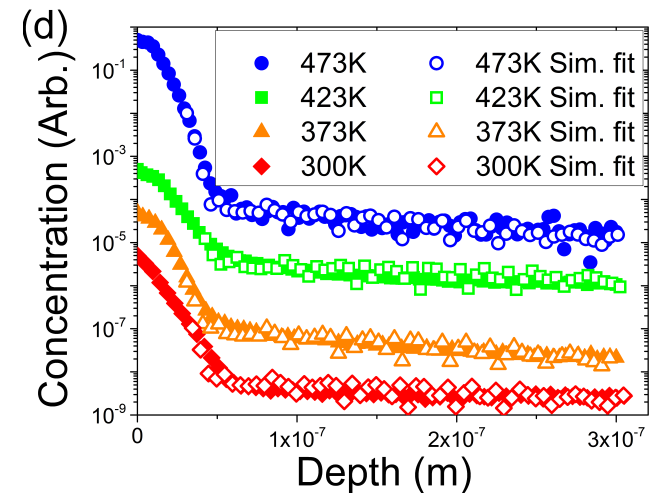
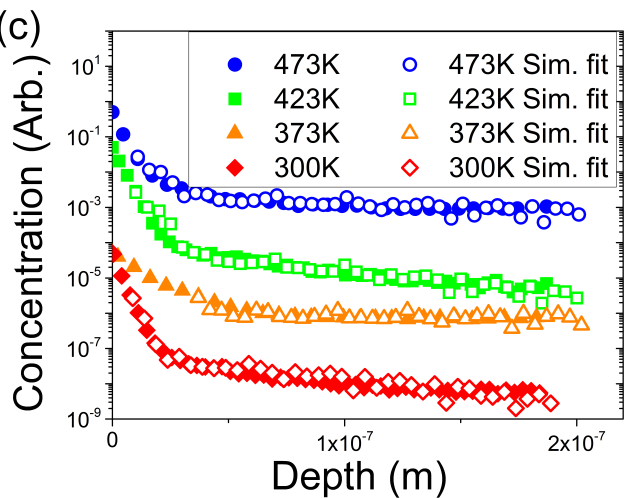
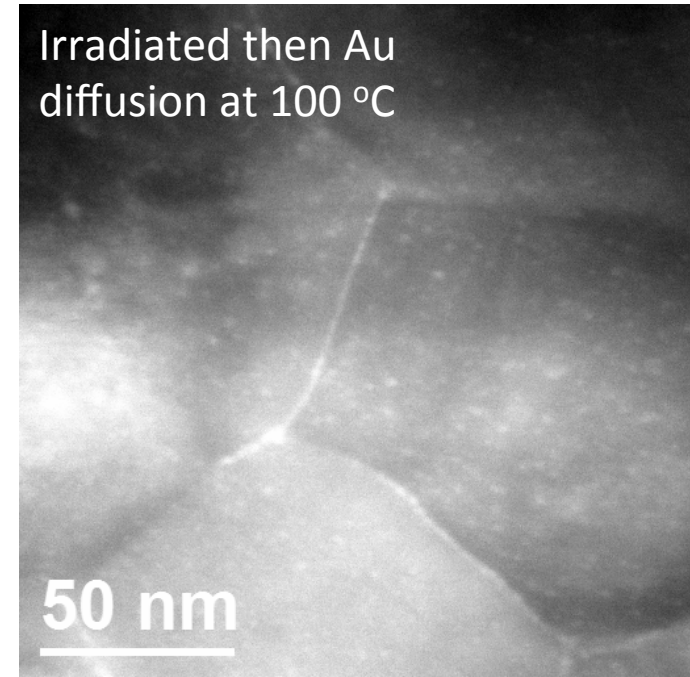
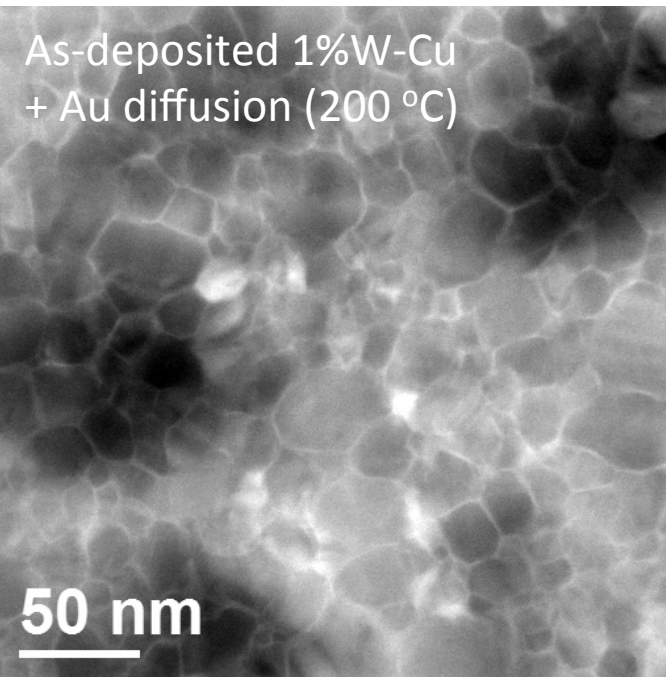


No Model

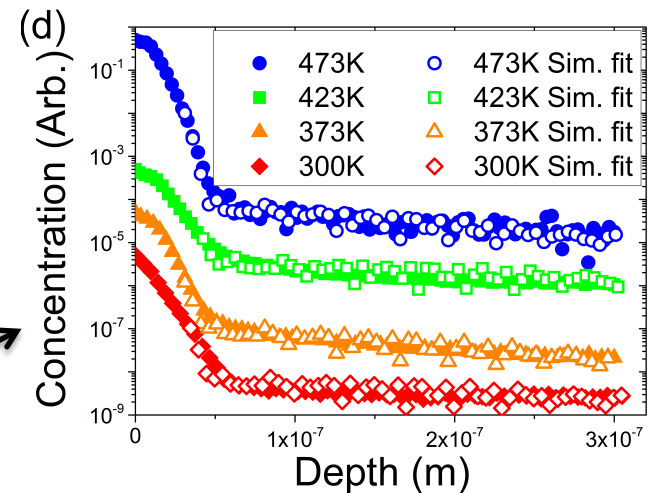
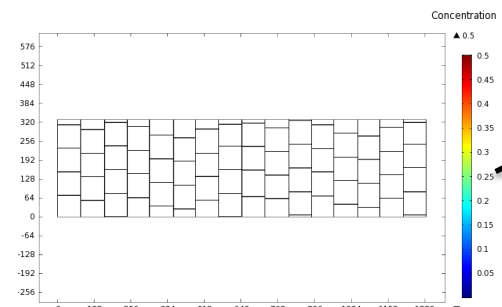
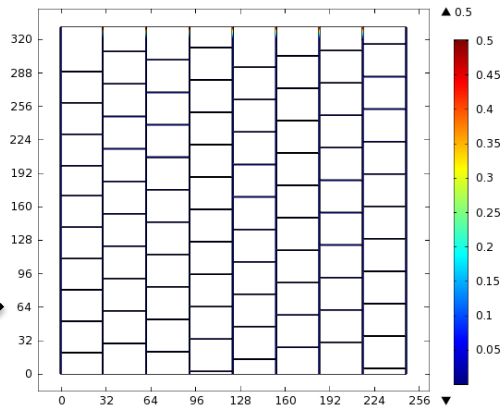
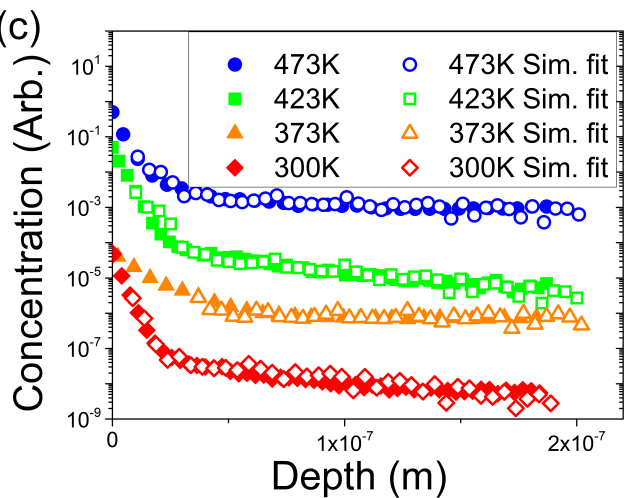
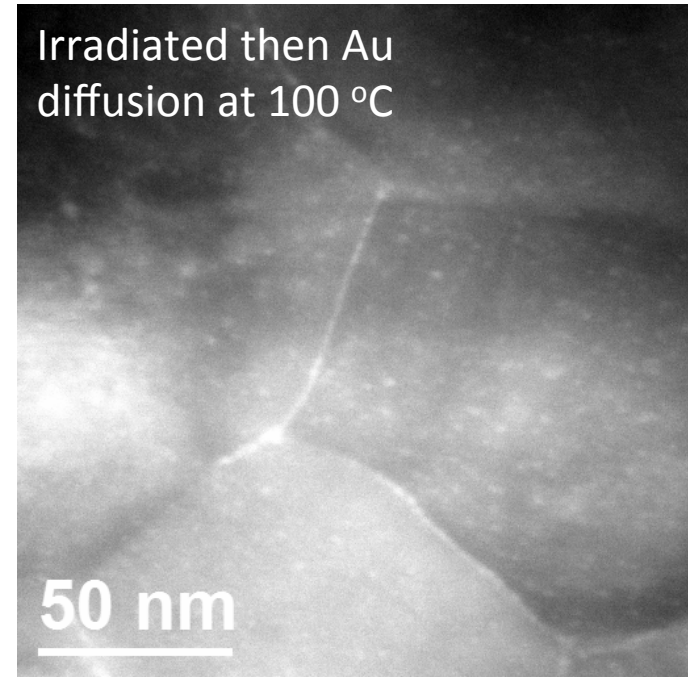
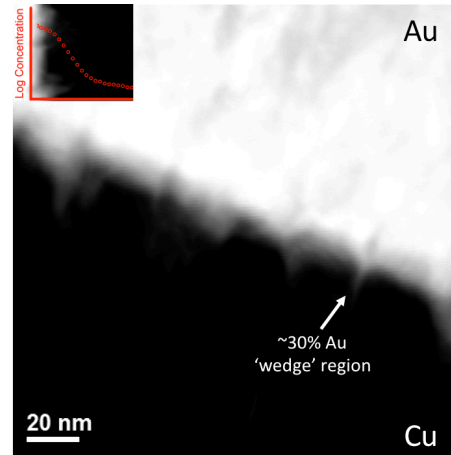
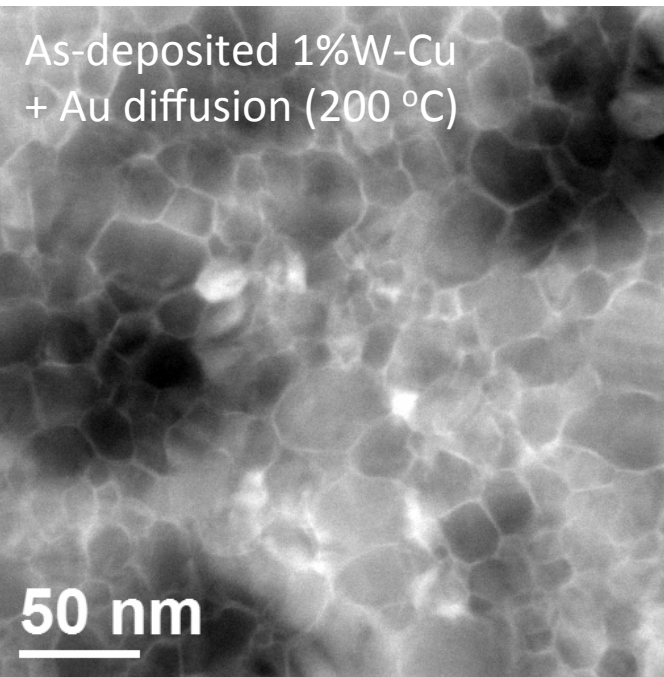


$$\delta D_b = v \left(\frac{\partial \ln \bar{c}}{\partial z} \right)^{-2}$$

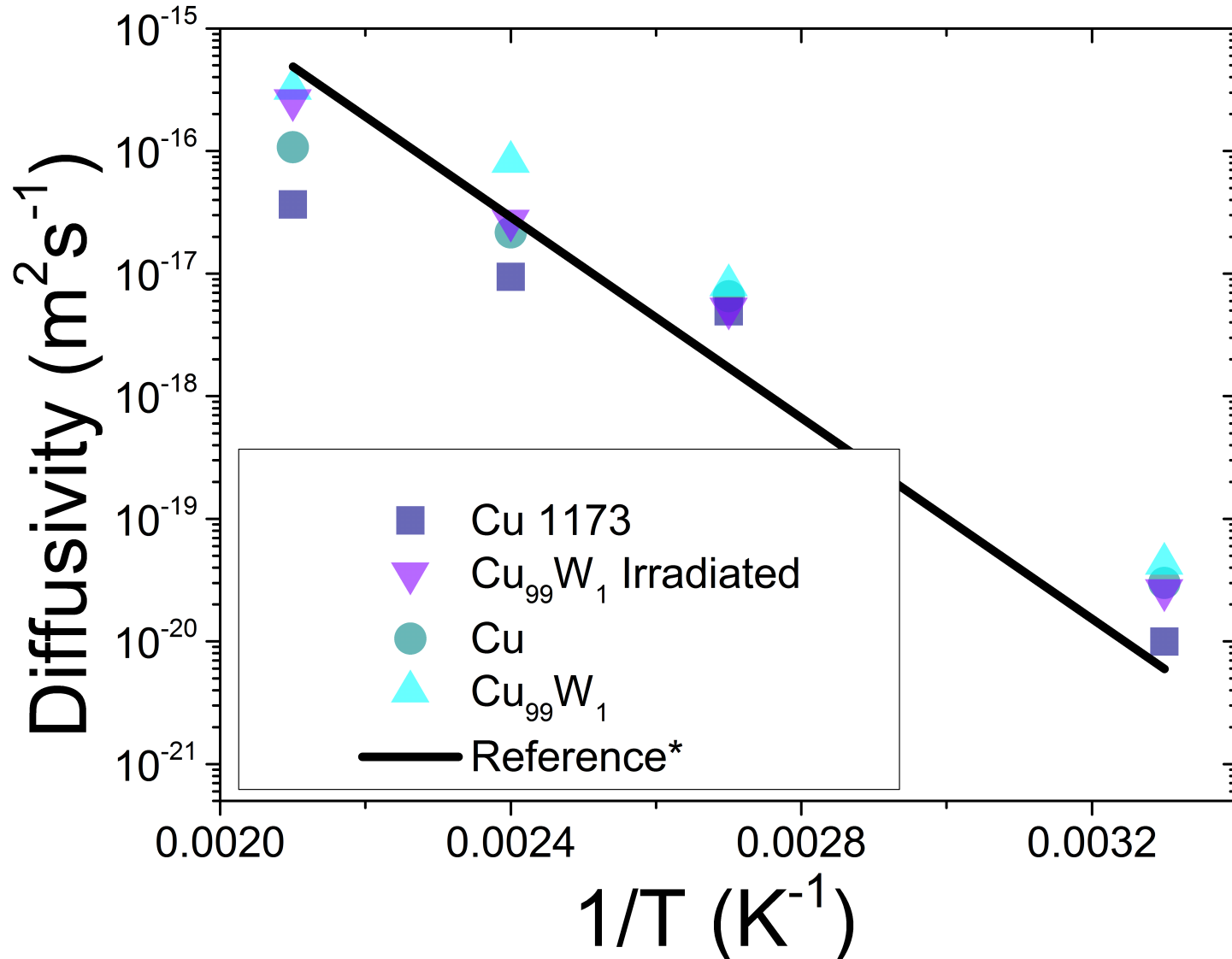
Au diffusion in Cu (Different G.S.'s)



Au diffusion in Cu (Different G.S.'s)



Results of Au G.B. Diffusion in Cu



Au diffusivity in the Cu – Limited ‘nano-effect’

Conclusions

- Significant entropy contribution, $\sim 4-8k$, associated with G.B. complexion transitions
- The excess entropy is manifest as enhanced diffusivity / G.B. mobility
- G.S. effect on diffusivity weak on thin film samples
- Thermal relaxations are qualitatively 'rapid' at temperatures where vacancy hops active